

# Focusing neutron optics with elastically bent perfect crystals

**Jiří Kulda**

Institut Laue-Langevin, Grenoble, France

# Acknowledgements

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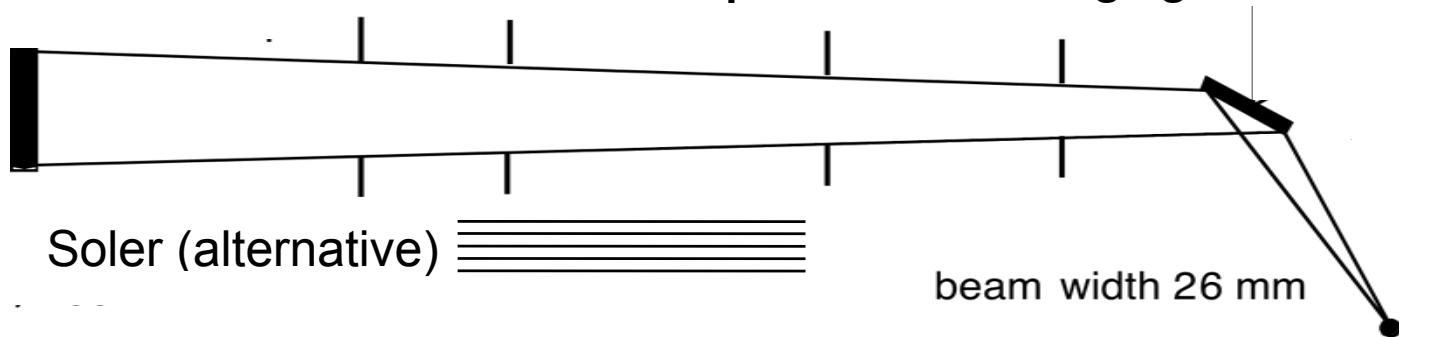
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P. Courtois, C. Menthonex	
J.-P. Vassali	ESRF
J.-M. Bisson, G. Pastrello	AZ Systemes, Grenoble

# Layout of the talk

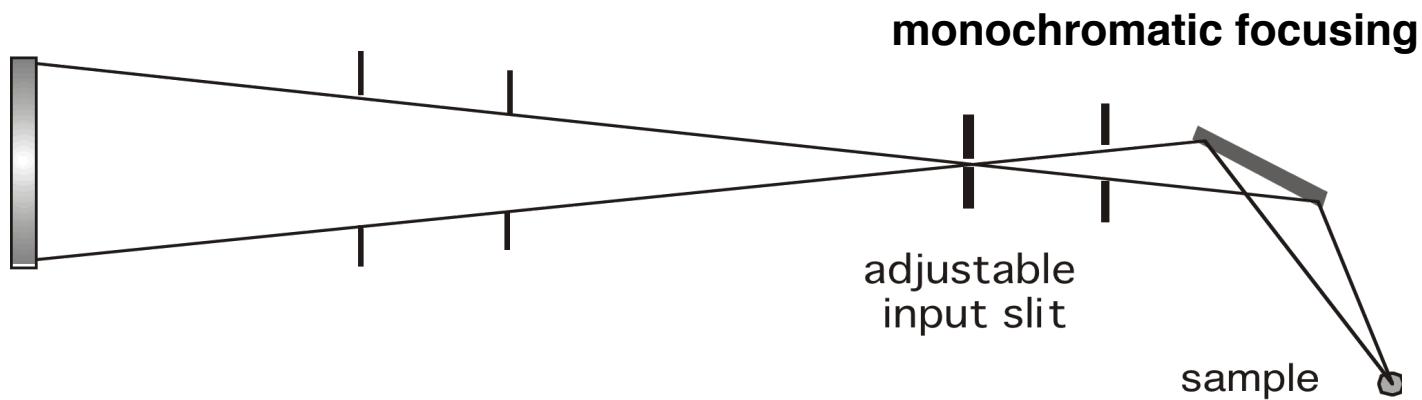
- 1. Optics**
- 2. Technology**
- 3. Applications**
  - 1. General TAS**
  - 2. Fine focusing reciprocal space**
  - 3. Fine focusing real space**
- 4. Conclusions**

# TAS layout

*"traditional" (... – 1999)*



*"modern" (2000 - ...)*



**intensity gain  $\approx 30x$**

**sample size  $> 1 \text{ cm}^3 \rightarrow 100 \text{ mm}^3$**

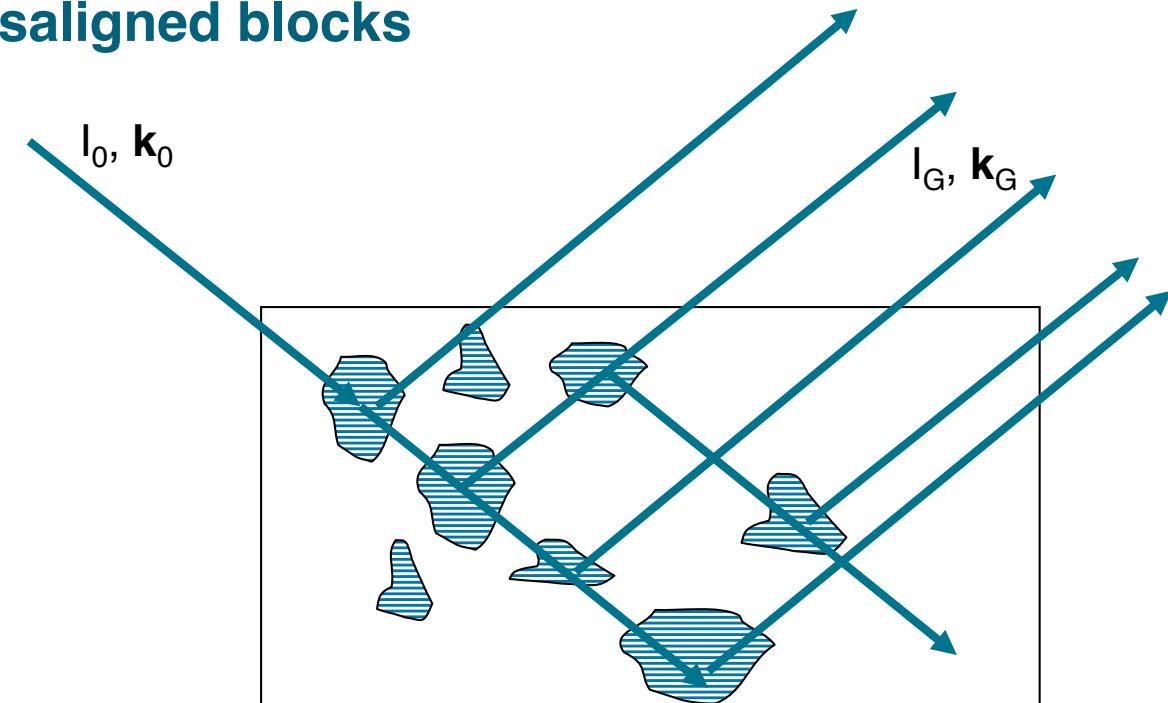
# Mosaic crystal

**assembly of (slightly) misaligned blocks**

Zachariasen's equations:

$$\partial I_0 / \partial s_0 = -\sigma(\varepsilon)(I_0 - I_G)$$

$$\partial I_G / \partial s_G = -\sigma(\varepsilon)(I_G - I_0)$$



$$\sigma(\varepsilon) = Q_{kin} w(\varepsilon) = \frac{F_G^2 \lambda^3}{v_0^2 \sin 2\theta} \frac{\exp[-(\varepsilon/\eta)^2]}{\eta \sqrt{\pi}}$$

kinematic reflectivity

mosaic distribution (Gaussian)

$$\varepsilon = \Theta - \Theta_0$$

angular deviation

# Mosaic crystal reflectivity

**Solutions to Zachariasen's eqs:**

**Bragg case:**

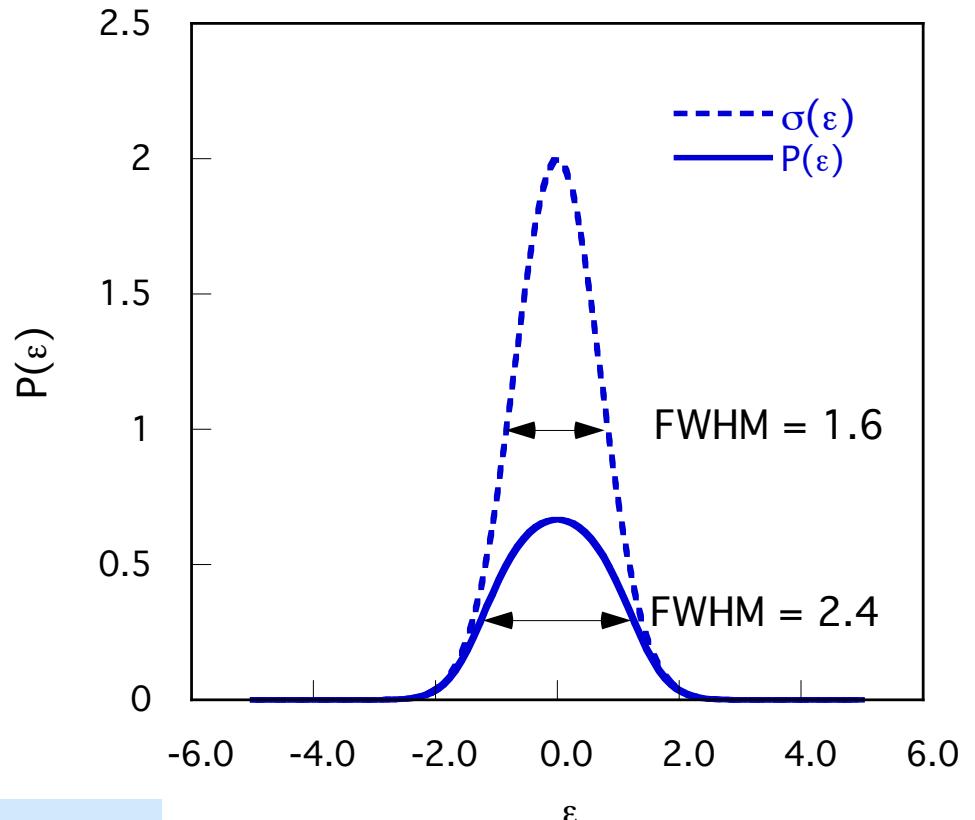
$$P(\varepsilon) = \frac{\sigma(\varepsilon)}{\mu + \sigma(\varepsilon) + u(\varepsilon) \coth ut}$$

$$P(\varepsilon) = \frac{\sigma(\varepsilon)}{\sigma(\varepsilon) + 1}, \quad \mu = 0$$

$$u = \sqrt{\mu(\mu + 2\sigma)}$$

**Laue case:**

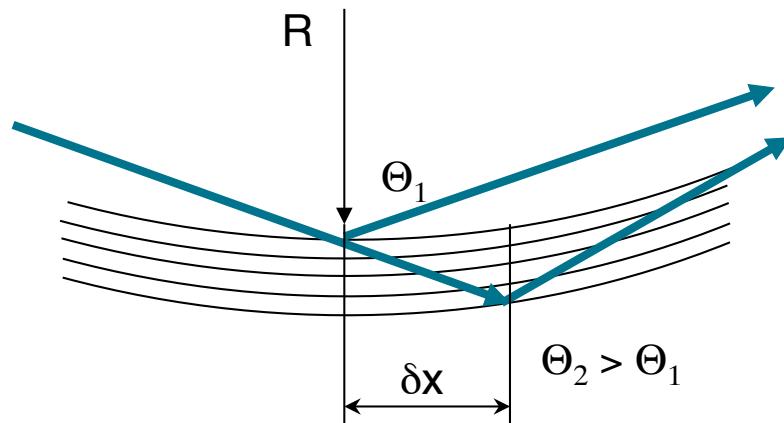
$$P(\varepsilon) = \exp[-(\mu + \sigma(\varepsilon))t] \sinh[\sigma(\varepsilon)t]$$



$$\sigma(\varepsilon) = Q_{kin} w(\varepsilon) = \frac{F_g^2 \lambda^3}{v_0^2 \sin 2\theta} \frac{\exp[-(\varepsilon/\eta)^2]}{\eta \sqrt{\pi}}$$

# Gradient crystals

## *Simple bending*

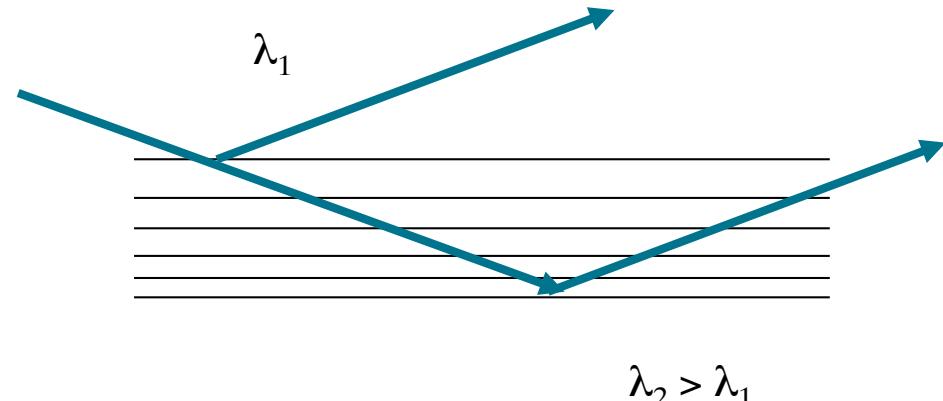


apparent mosaic width:

$$\delta\Theta = \frac{\delta x}{R}$$

wavelength band:

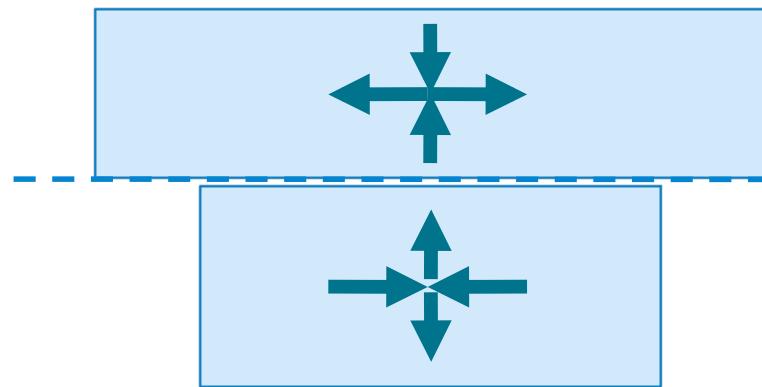
$$\Delta\lambda = 2\Delta d_{hkl} \sin\Theta$$



*Simple  $d_{hkl}$  gradient)*

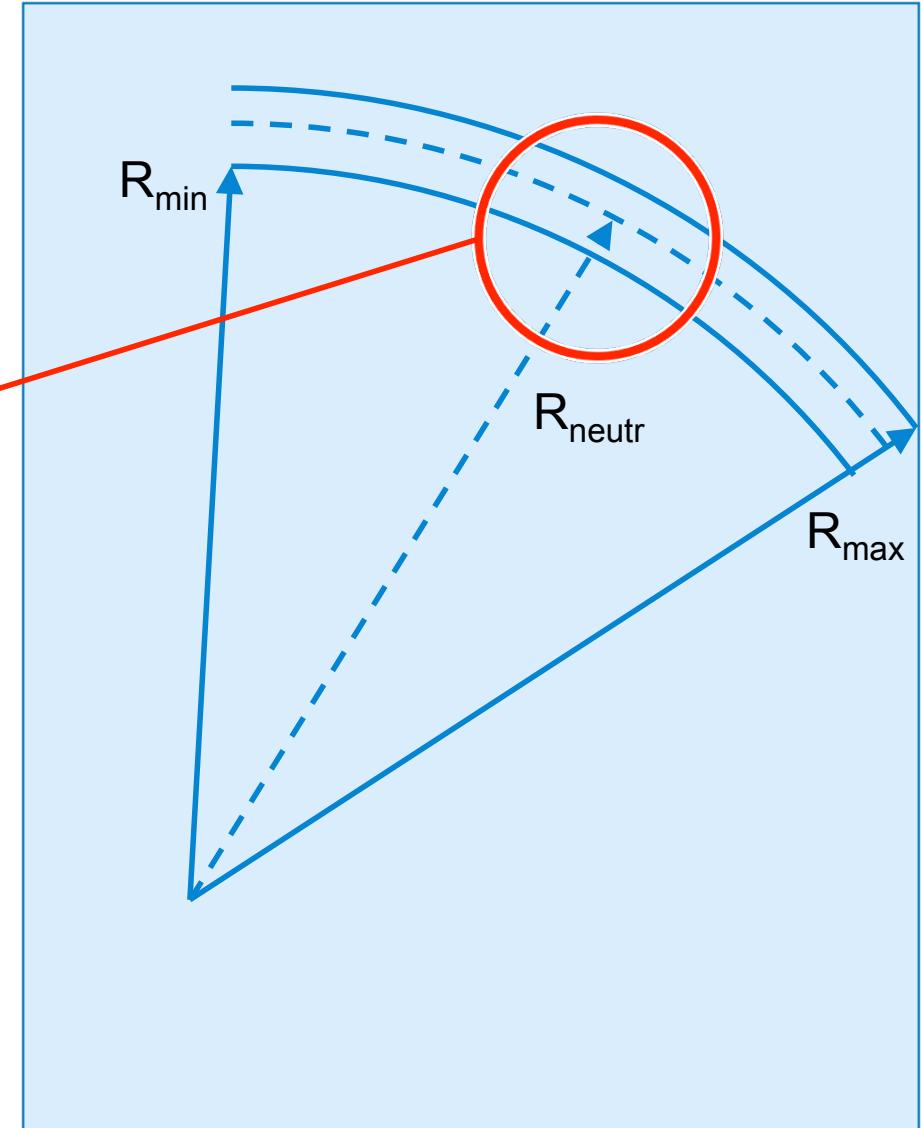
# Gradient crystals

## Elastically bent crystal



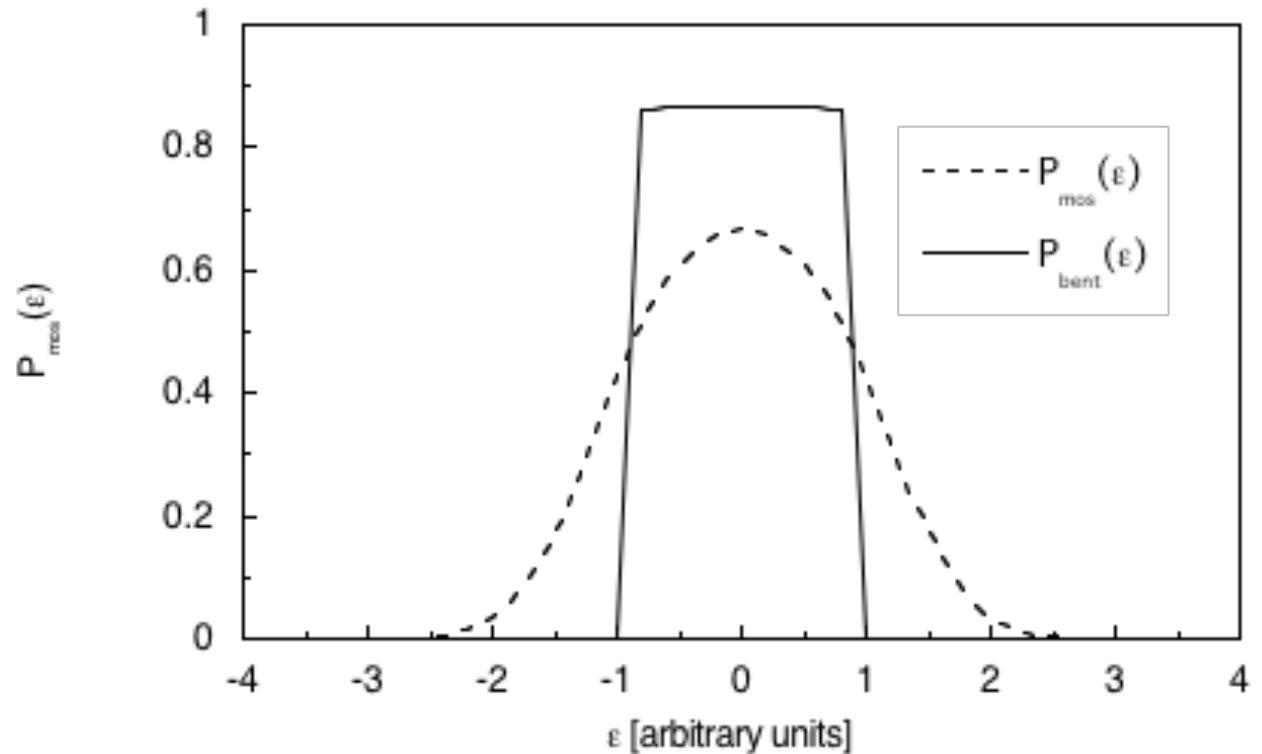
Effective bending radius variation  
due to strain/stress equilibrium

analytically solvable  
(cf. eg. Landau & Lifshitz)



# Gradient vers. mosaic

## Reflection profile



## Integrated reflectivity

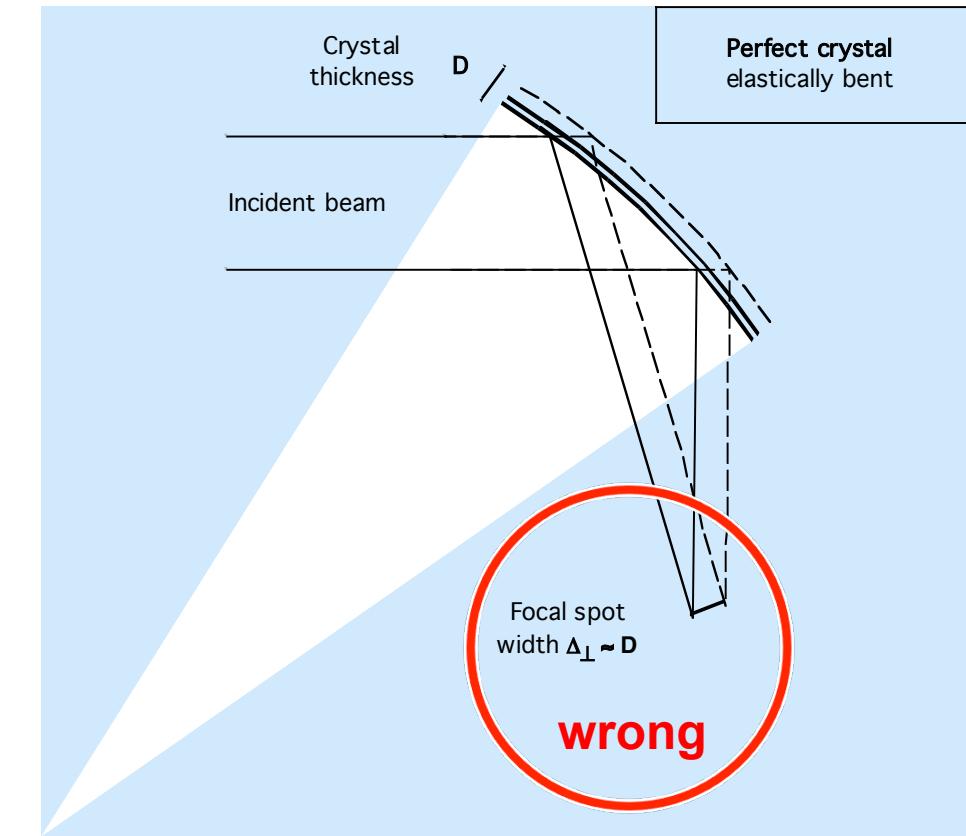
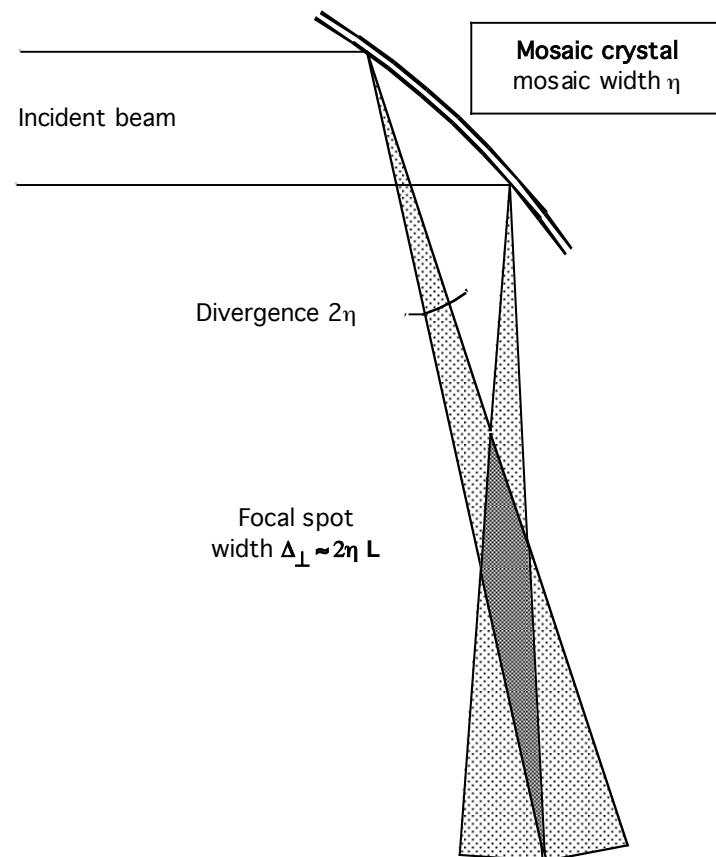
$$\rho(\Theta) = \frac{\Delta x}{R} \left[ 1 - \exp\left( \frac{Q_{\text{kin}} R}{\cos \Theta} \right) \right]$$

J. Kulda, Acta Cryst. A40 (1984) 120

- almost rectangular rocking curve with minimum tails
- smaller crystal thickness required to achieve given reflectivity

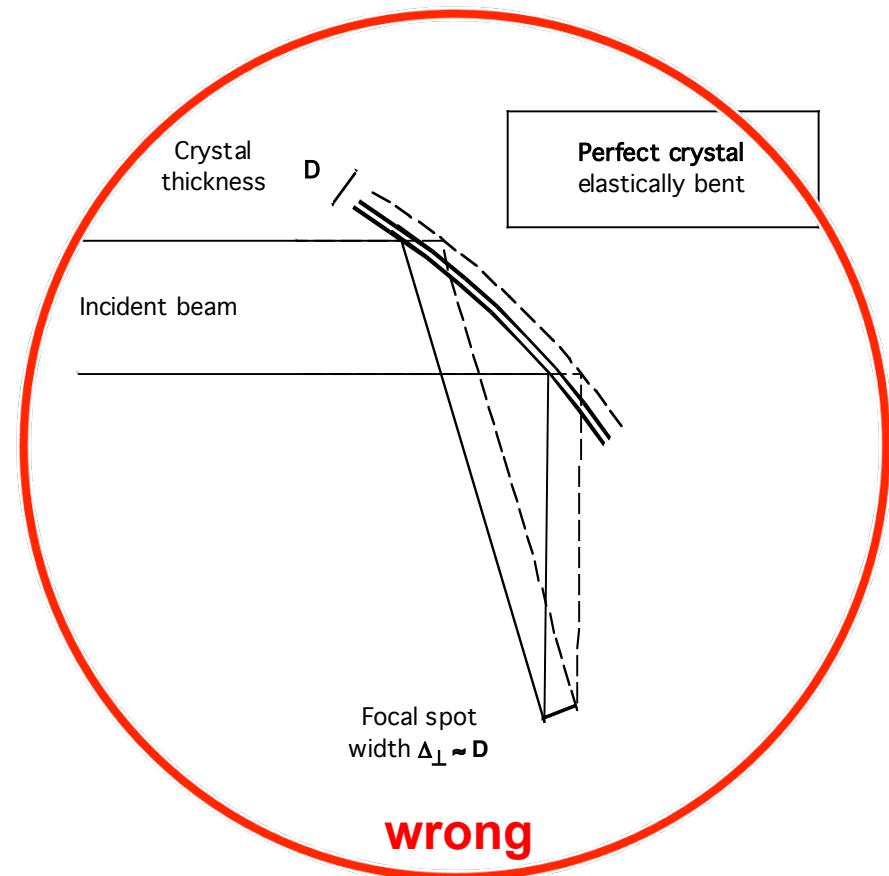
# Focusing properties

## real space

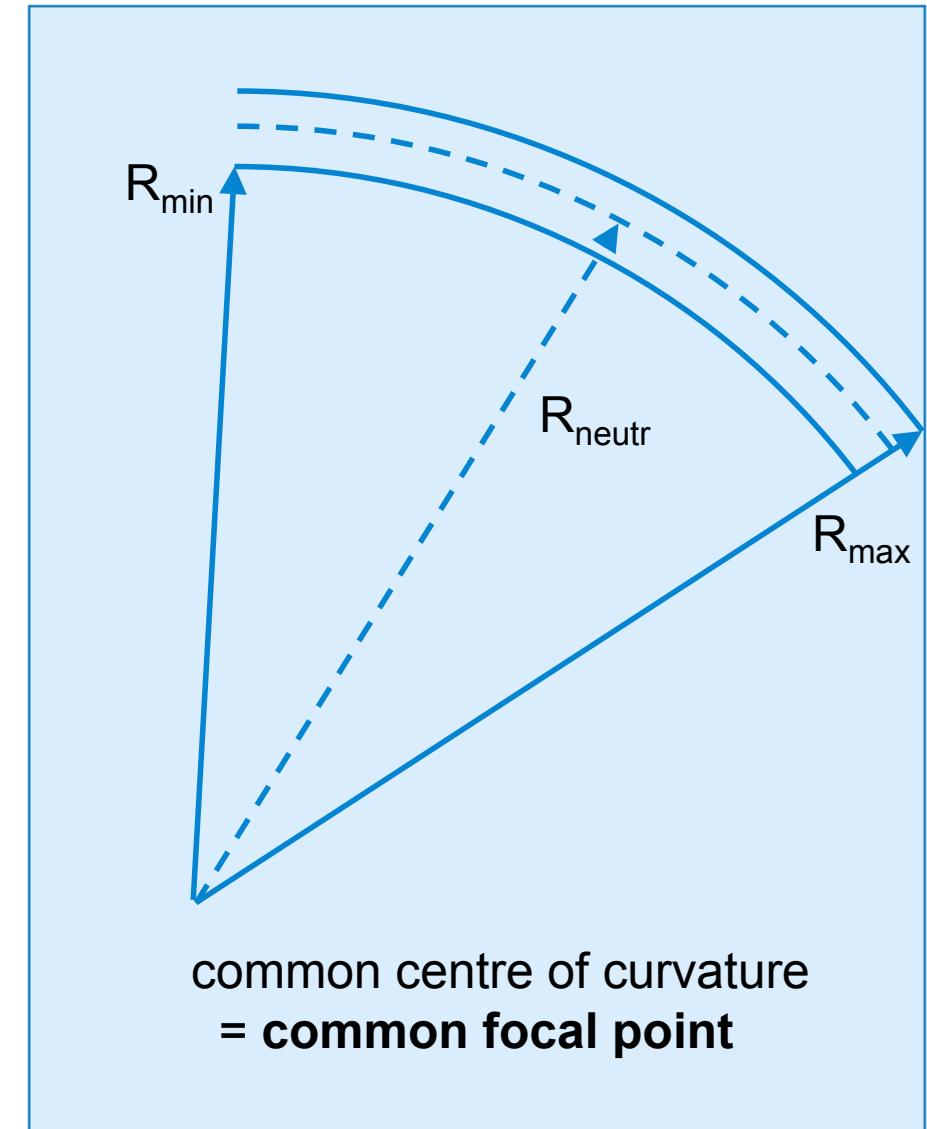


Kulda & Saroun, Nucl. Inst. Meth. A379 (1996) 155

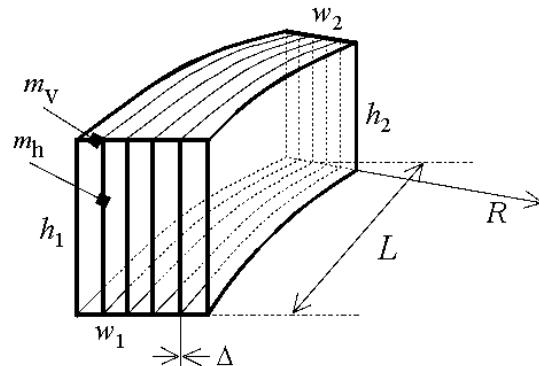
# Gradient crystals



Kulda & Saroun, Nucl. Inst. Meth. A379 (1996) 155

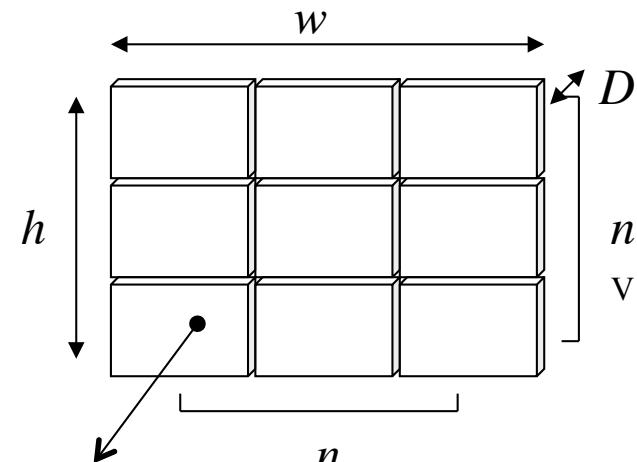
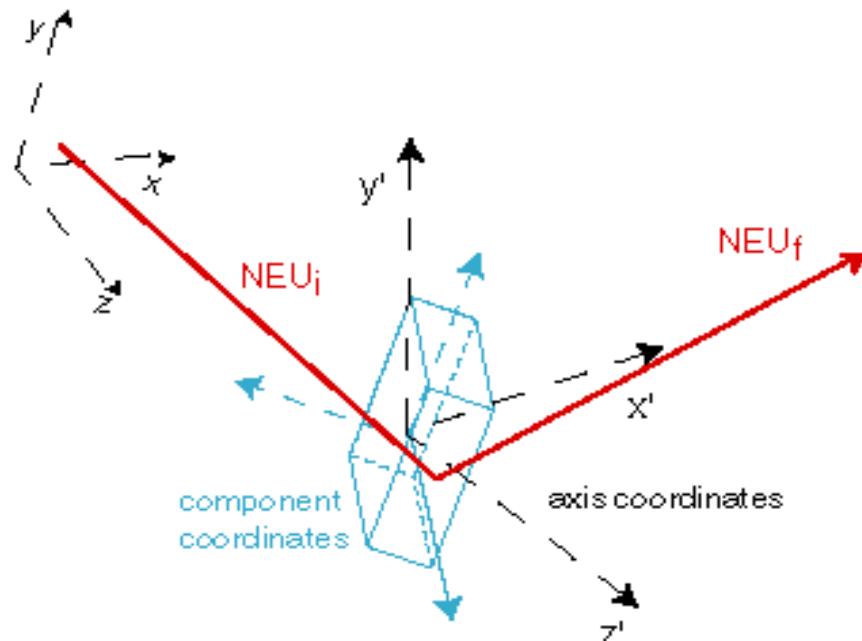


# RESTRAX



- neutron ray-tracing or multi-Gaussian convolution
- diffraction/reflection optics of neutron instruments
- realistic crystal description (mosaic, elastically bent)
- highly optimized F77/F95 code

<http://omega.ujf.cas.cz/restrax>



$\eta_H, \eta_V, \chi, d_{hkl}^H, Q_{hkl}, \mu, \dots$

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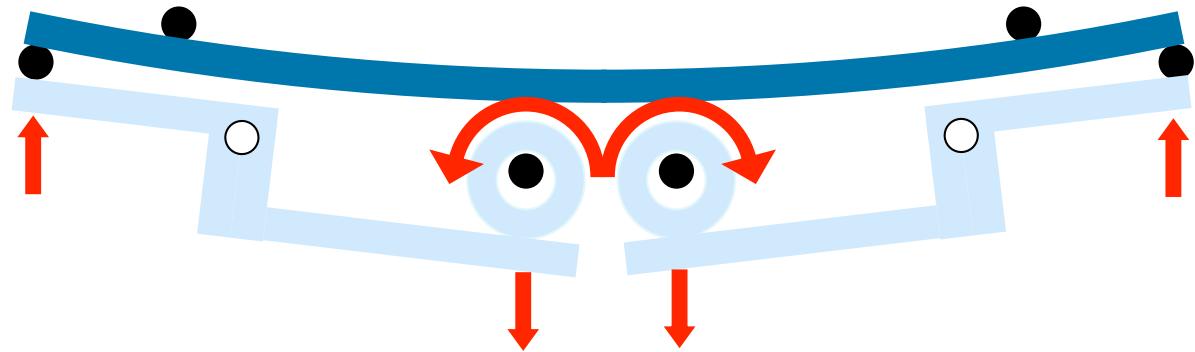
# Si bender - function scheme

- **horizontal focusing:** four-point bending device



$$R_h = \text{inf.}$$

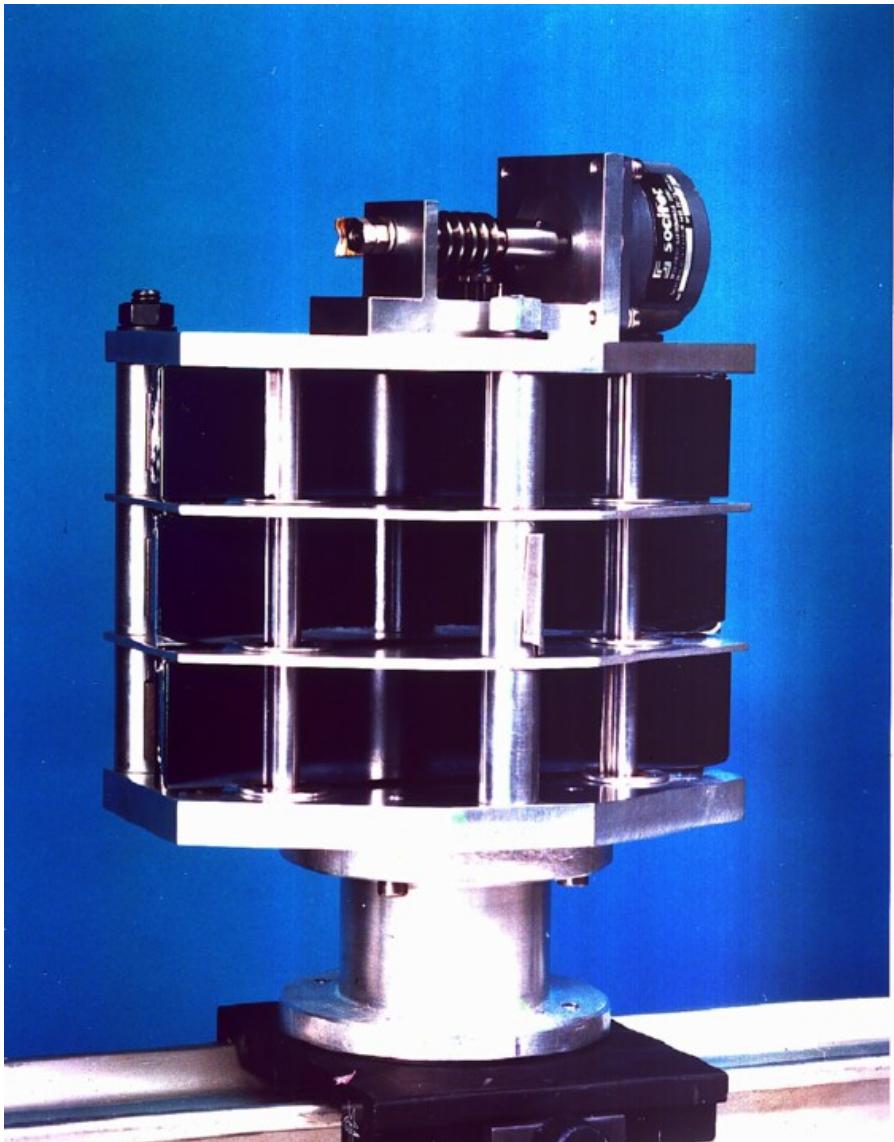
$$R_h = 1 - 20 \text{ m}$$



- **vertical focusing:** inclining segments & bell-shaped cams

$$R_v = 0.2 \text{ m} - \text{inf.}$$

# Si bender - 1<sup>st</sup> generation



## 1st generation:

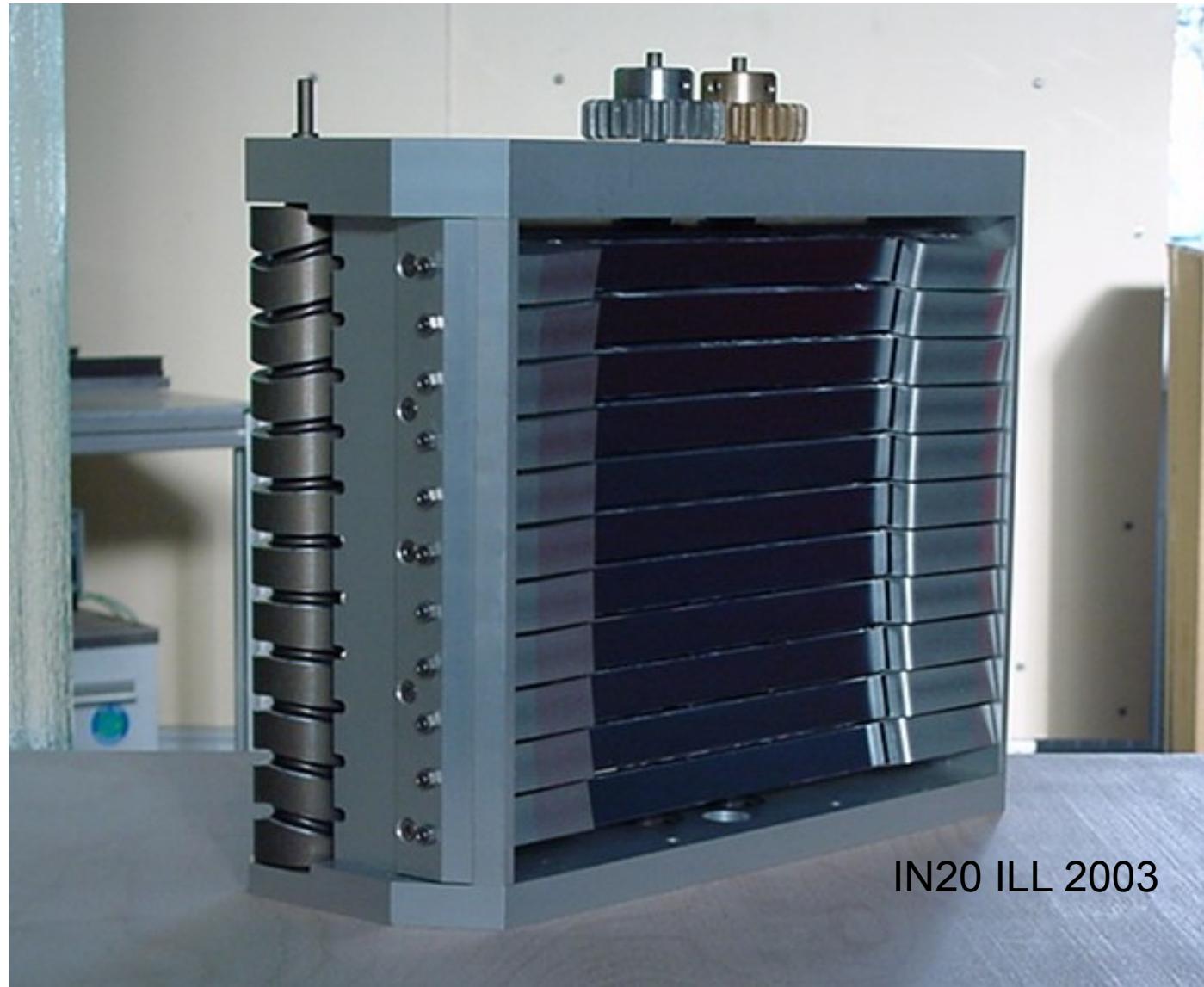
- variable horizontal curvature
- fixed vertical curvature
- 3 vertical segments (40 mm)
- blade thickness 3-5 mm
- active length 120 mm



## 2nd generation:

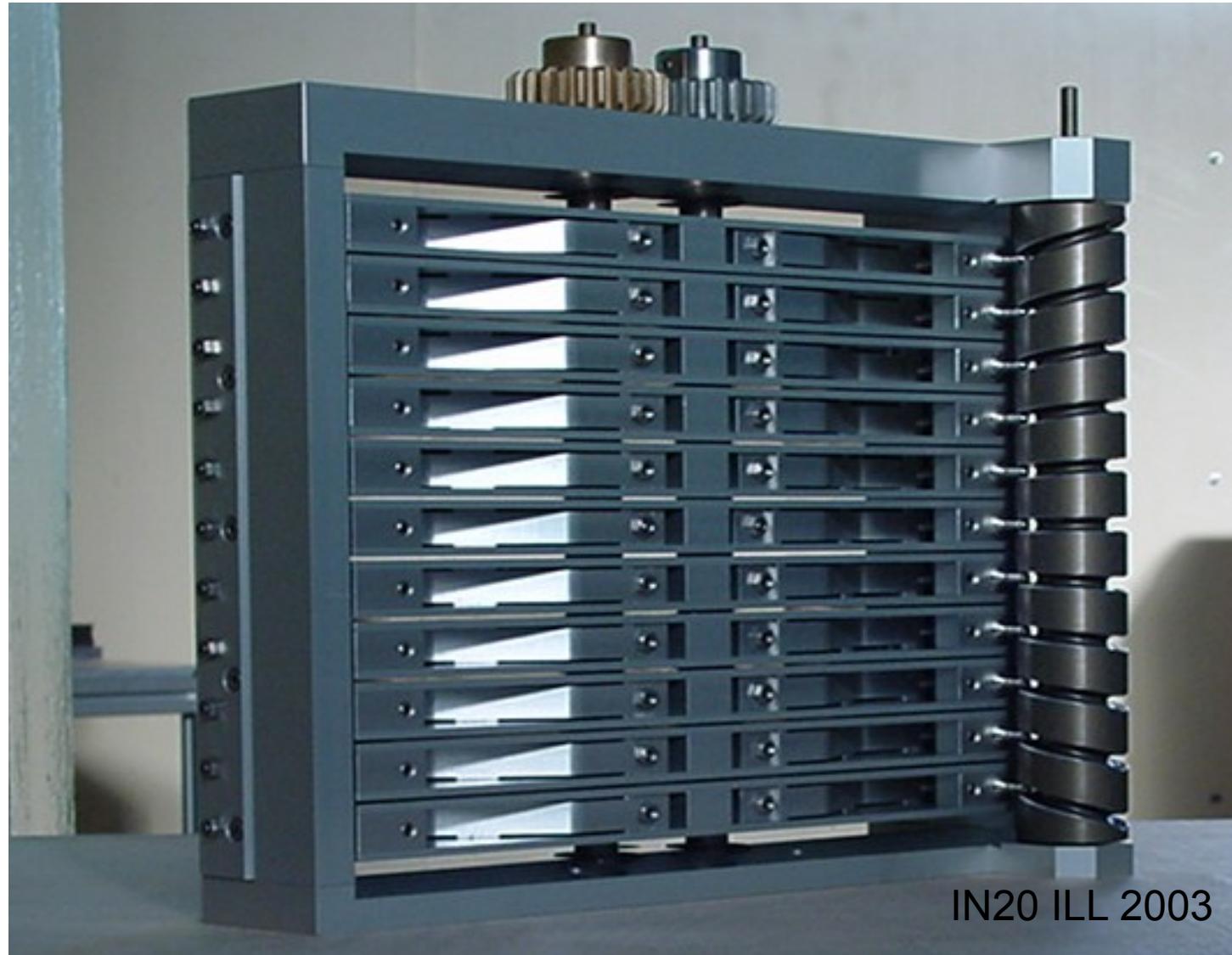
- *variable horizontal AND vertical curvature*
- *segment height < 20 mm*
- blade thickness < 1 mm  
(> 10 per pack)

# Si bender - front



IN20 ILL 2003

# Si bender - back

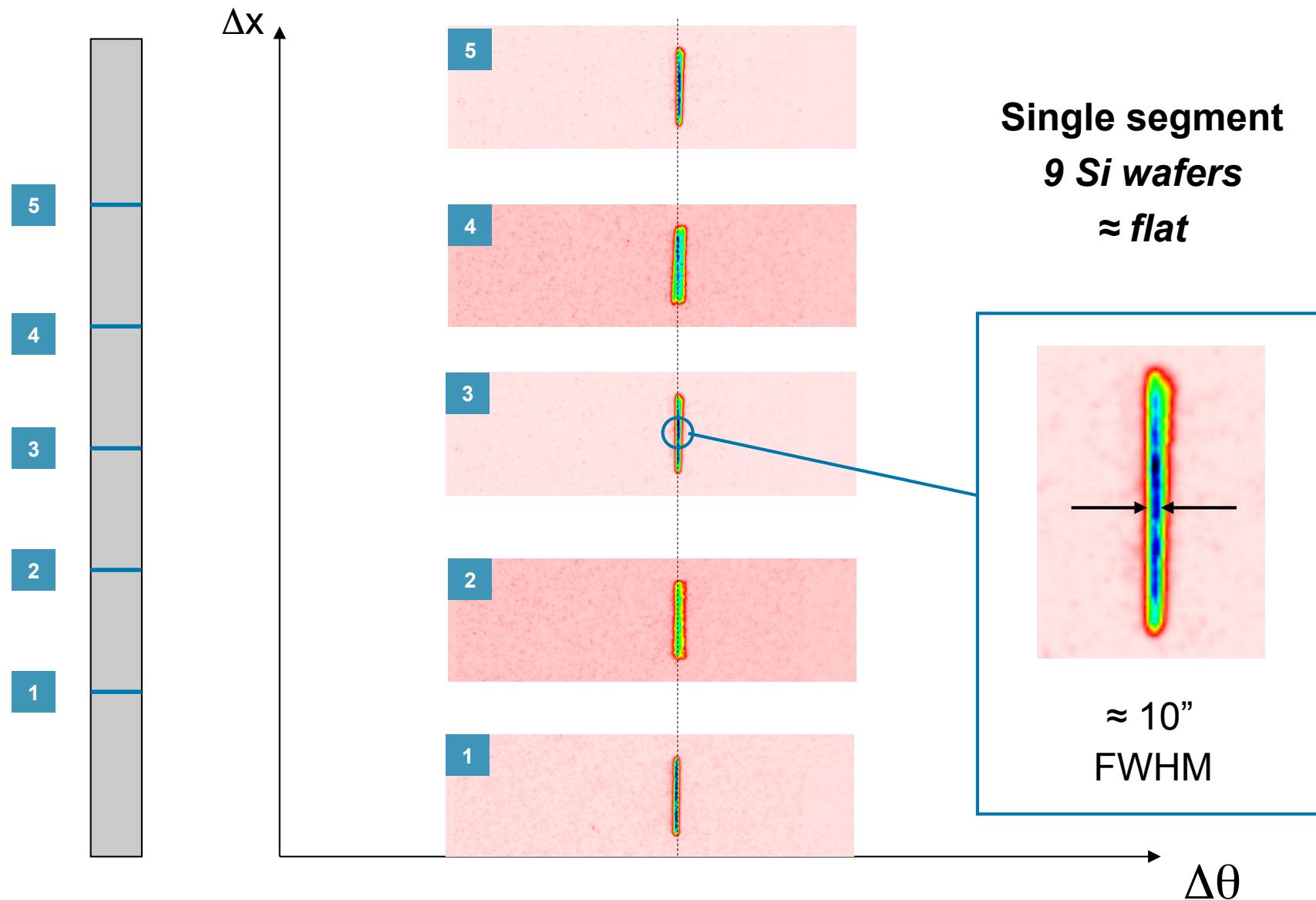


# Si wafers

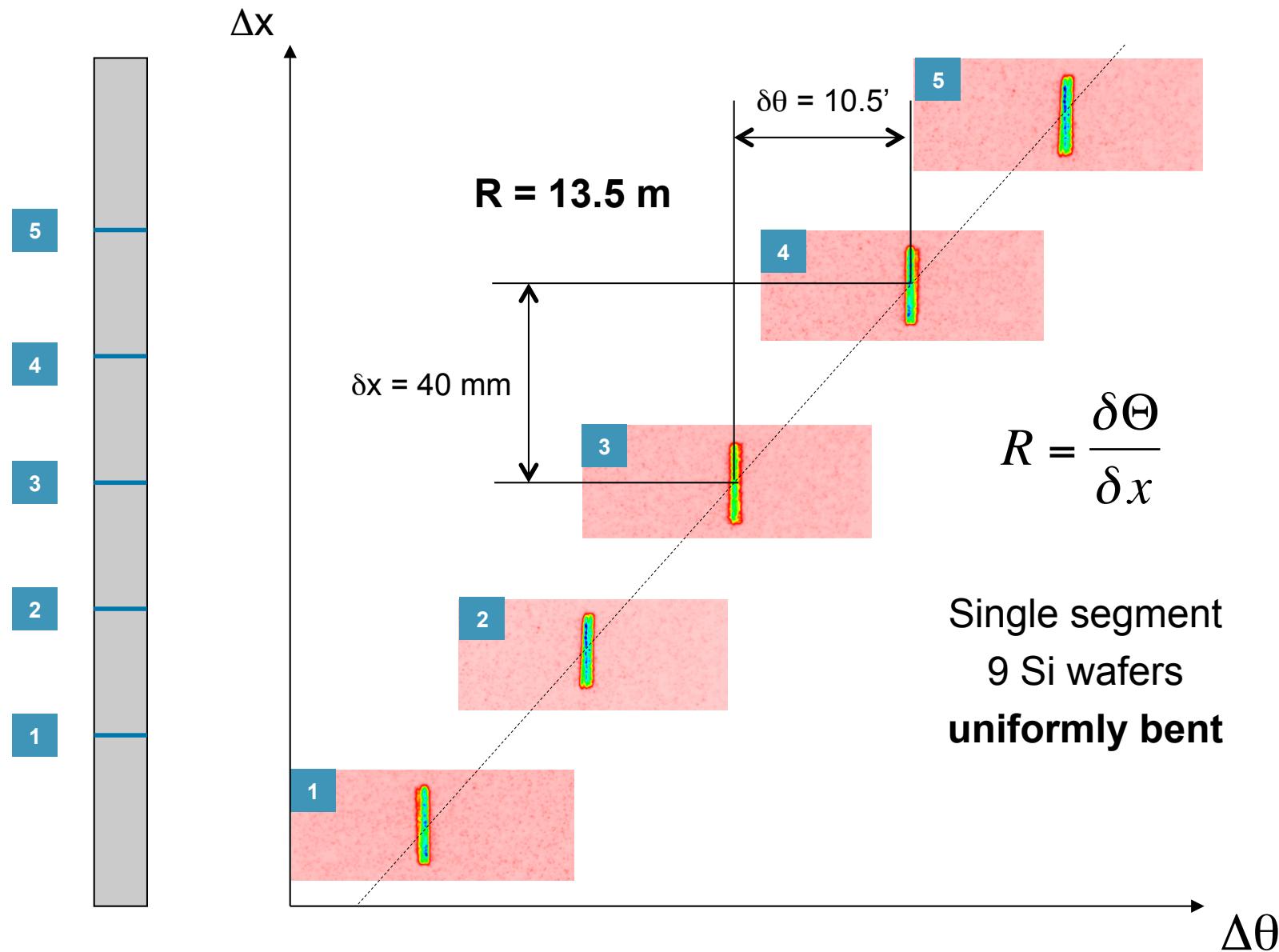
- 99 wafers (11 segments, 9 wafers each)
- size  $265 \times 17 \times 1 \text{ mm}^3$
- largest face (111)
- surface as-cut (multiwire saw!) & etched



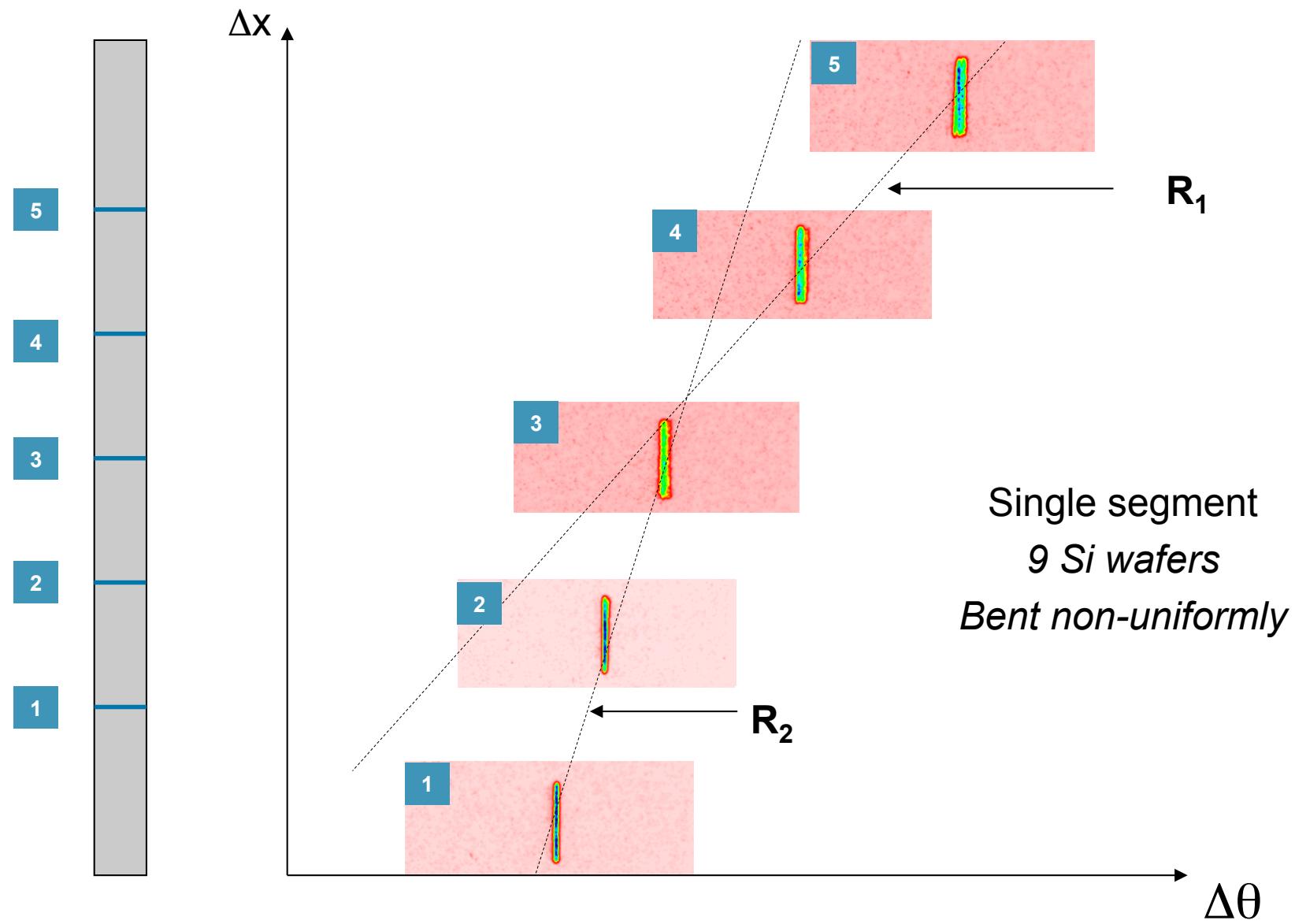
# X-ray tests (I)



## X-ray tests (II)



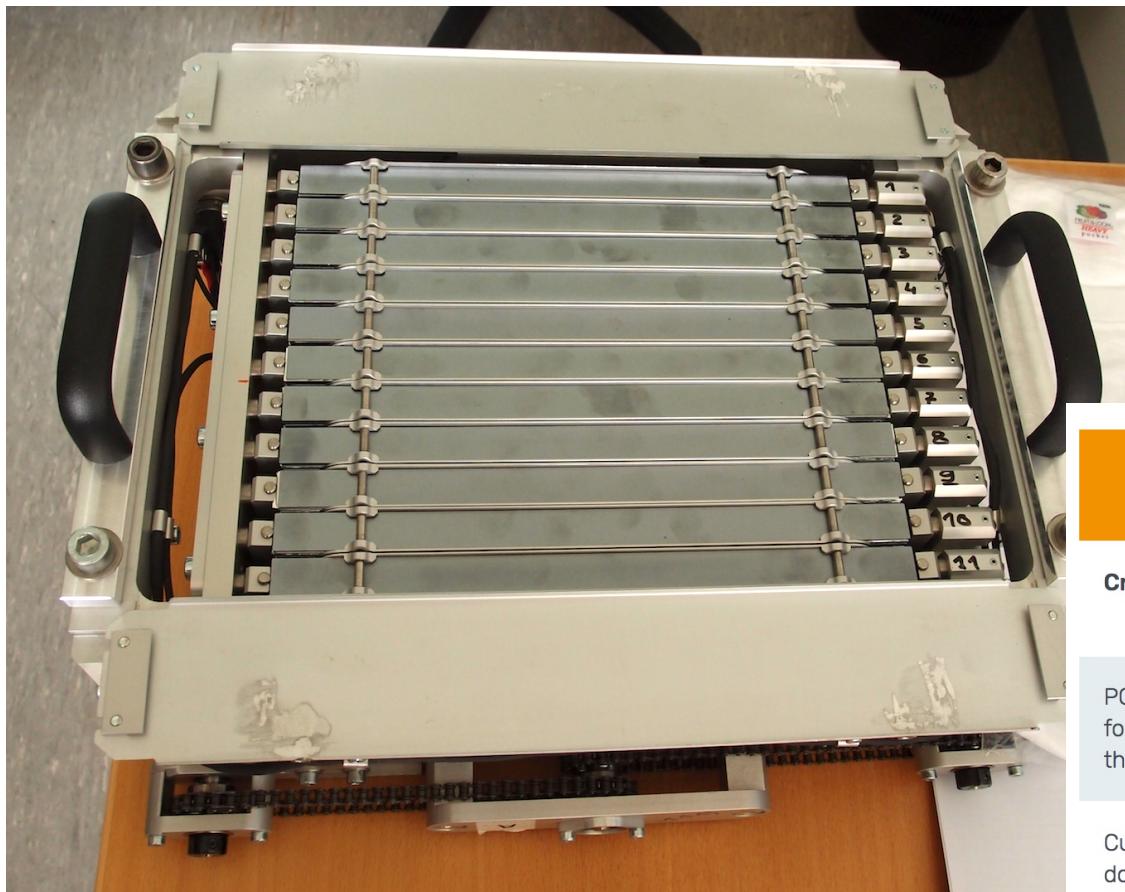
# X-ray tests (III)



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# Si (111) vers. PG(002)



monochromatic flux  $\approx 1/3$  PG

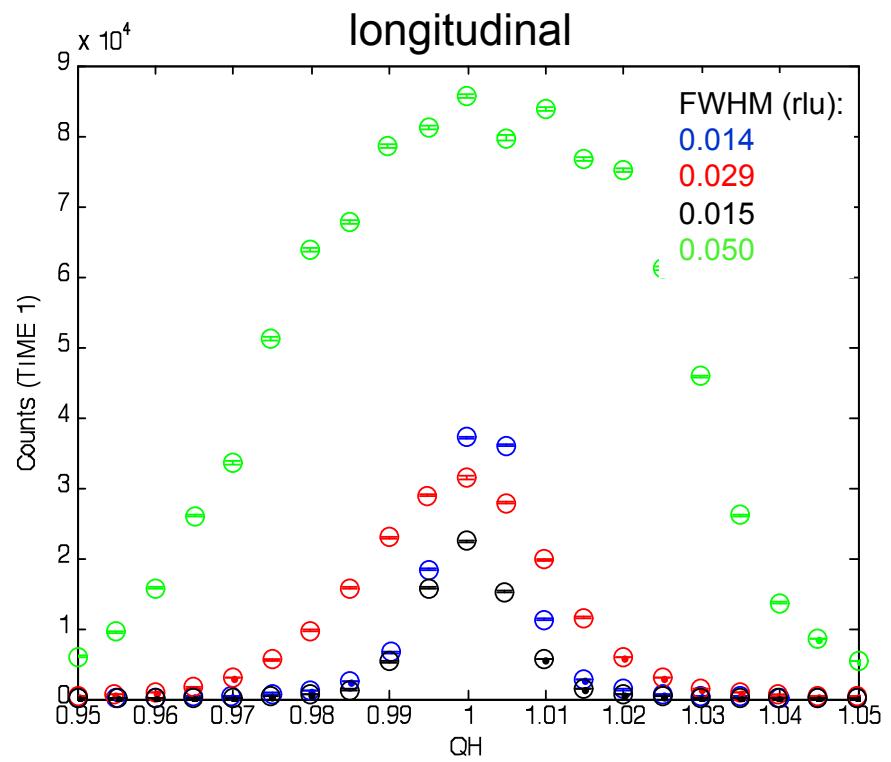
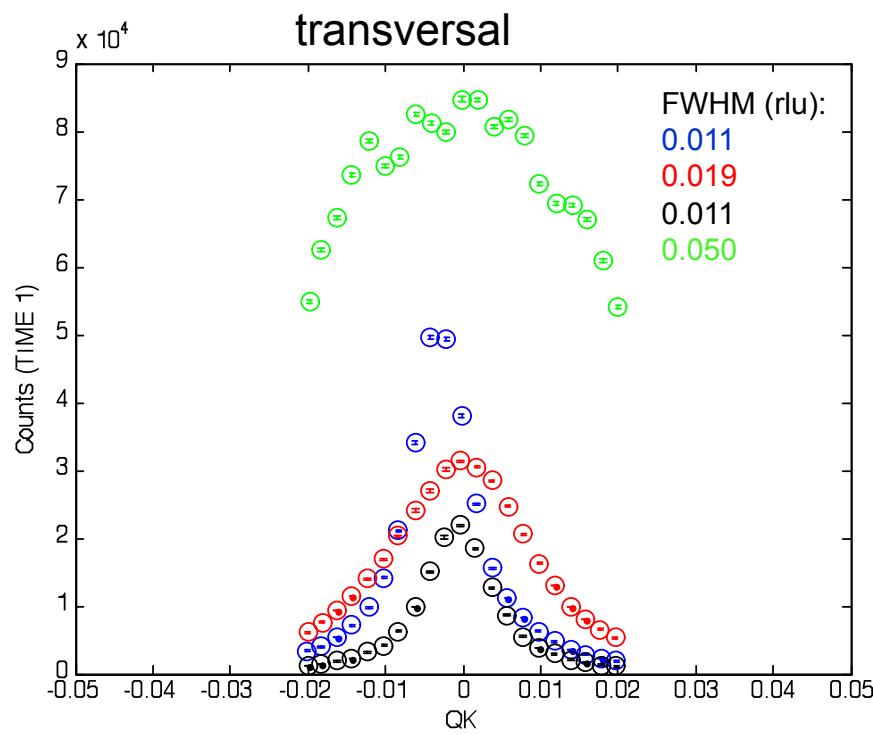
**IN1, IN8, IN20, ThALES**

## Monochromator

Crystal	W x H (mm <sup>2</sup> )	k <sub>i</sub> /Å <sup>-1</sup>	flux/10 <sup>8</sup> n cm <sup>-2</sup> s <sup>-1</sup>
PG (002) double focusing, three faces	233x197	2.662 4.1	2.0 6.5
Cu (200) variable double-focusing, anisotropic mosaic (h:25', v:10')	233x197	4.1 7.0	4.6 3.0
Si (111) bent perfect crystals, fixed horiz. curvature optimized for k=3.5Å <sup>-1</sup>	180x197	2.662 4.1	0.8 3.4

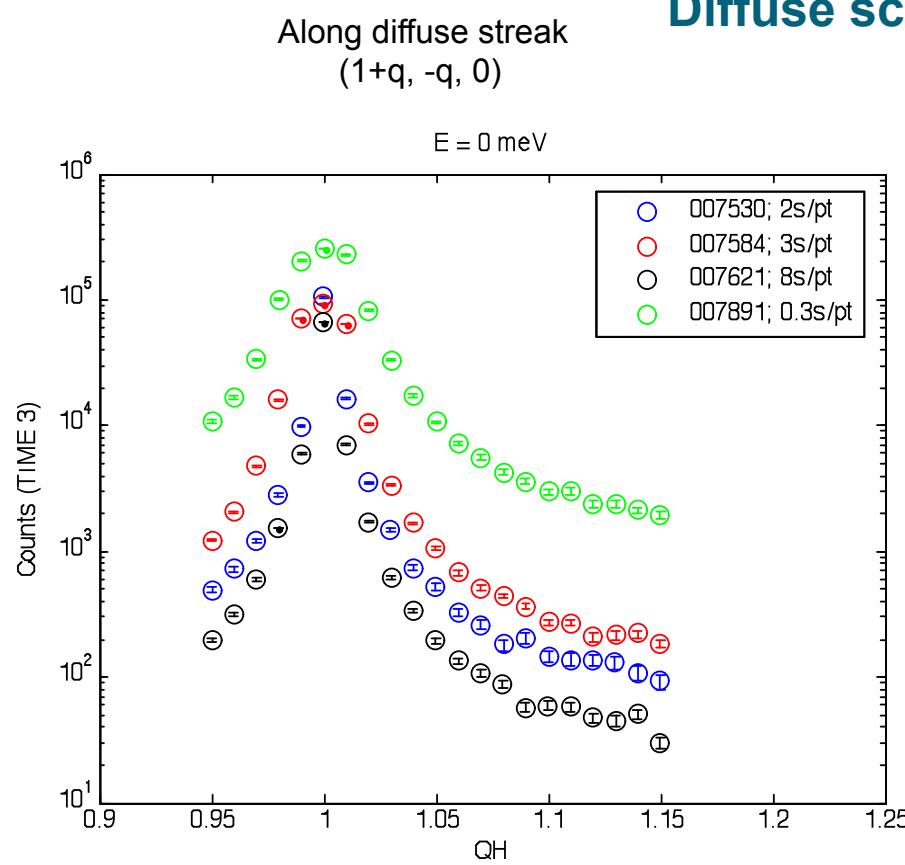
# Si (111) vers. PG(002)

## Bragg width (PMN 100)

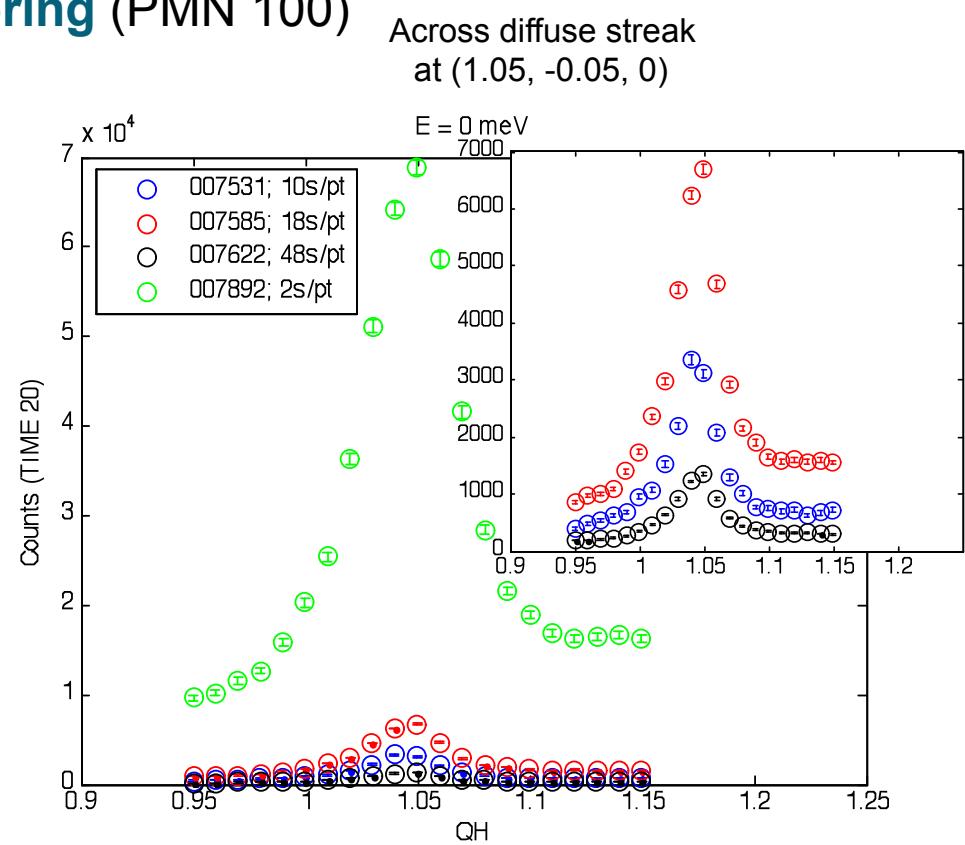


PG-PG	open	DTR 40
PG-PG	40' – 40'	DTR 40
Si-Si	open	DTR 10
Si-Si	40' – 40'	DTR 40

# Si (111) vers. PG(002)



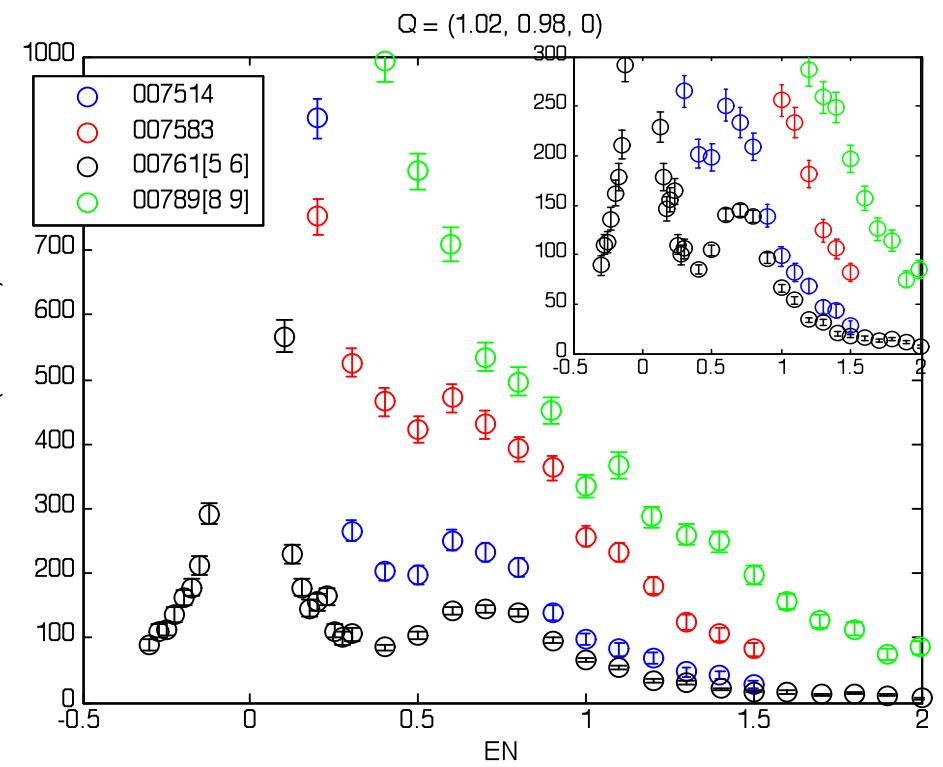
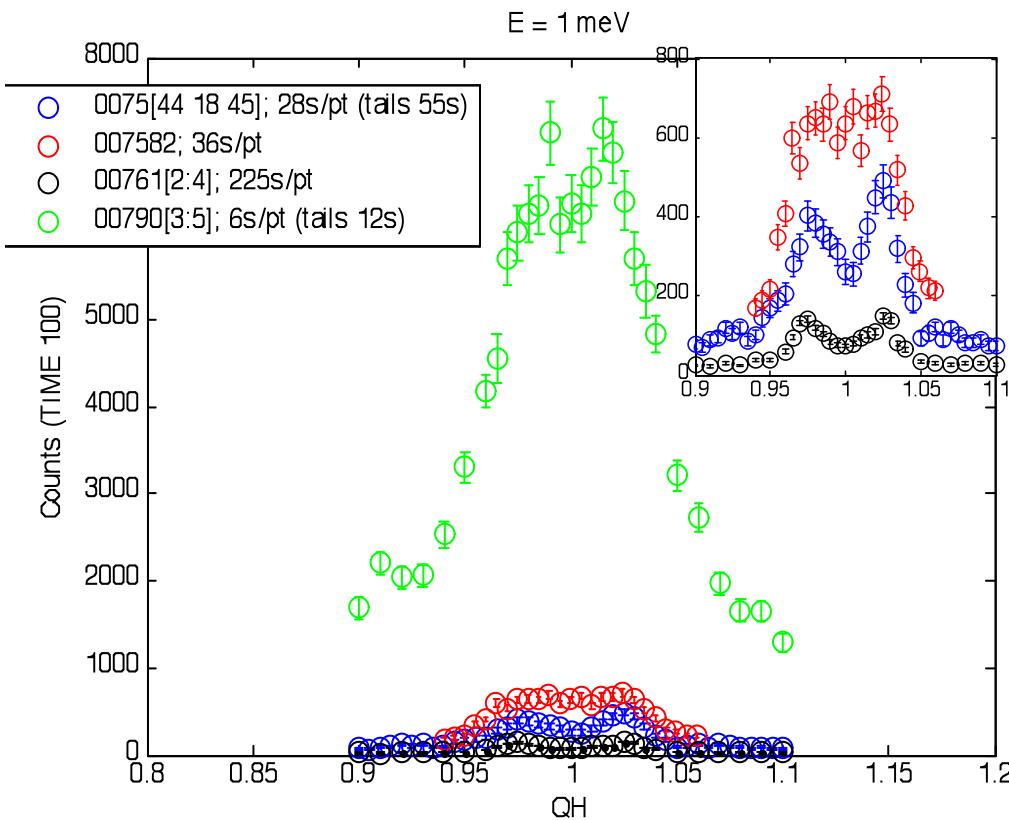
## Diffuse scattering (PMN 100)



PG-PG	open	DTR 40
PG-PG	40' – 40'	DTR 40
Si-Si	open	DTR 10
Si-Si	40' – 40'	DTR 40

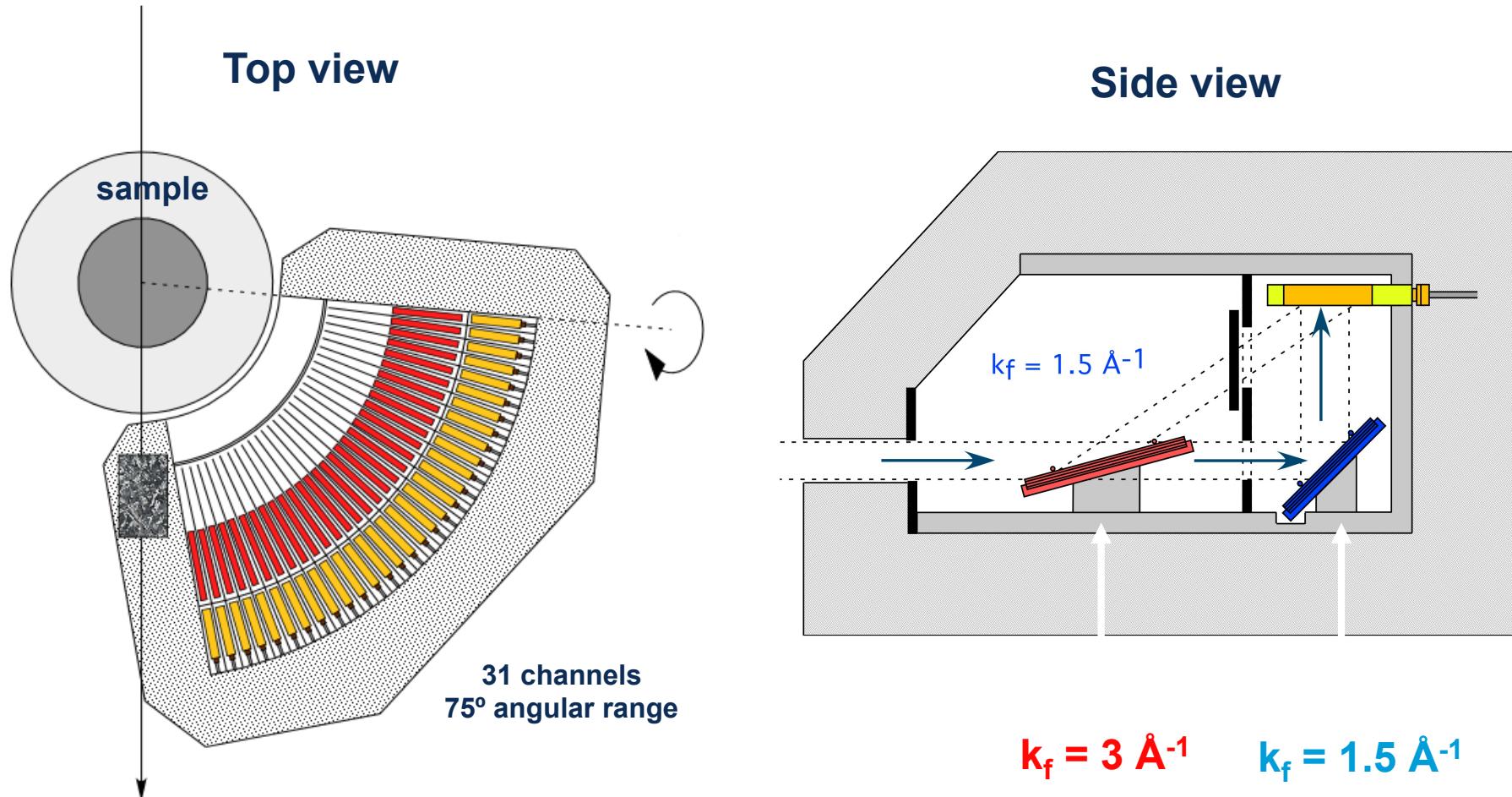
# Si (111) vers. PG(002)

## Phonons & QE signal (PMN 110)



PG-PG	open	DTR 40
PG-PG	40' – 40'	DTR 40
Si-Si	open	DTR 10
Si-Si	40' – 40'	DTR 40

# FlatCone multianalyzer





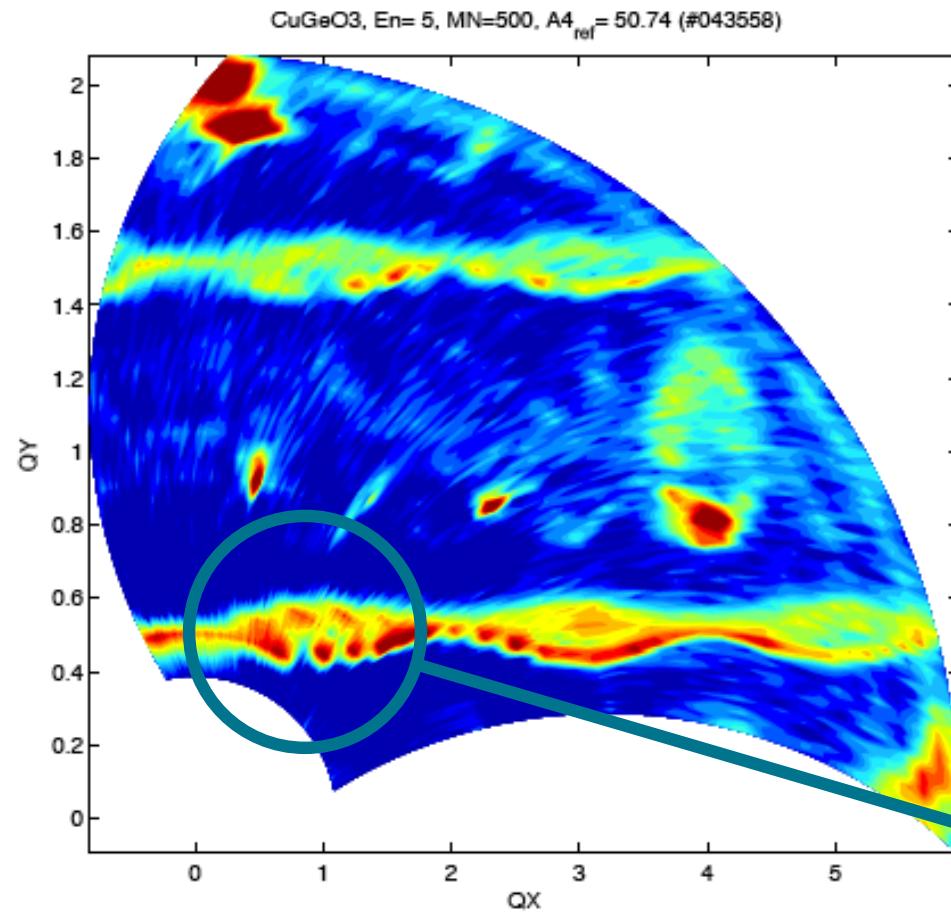


IN20, September 2006

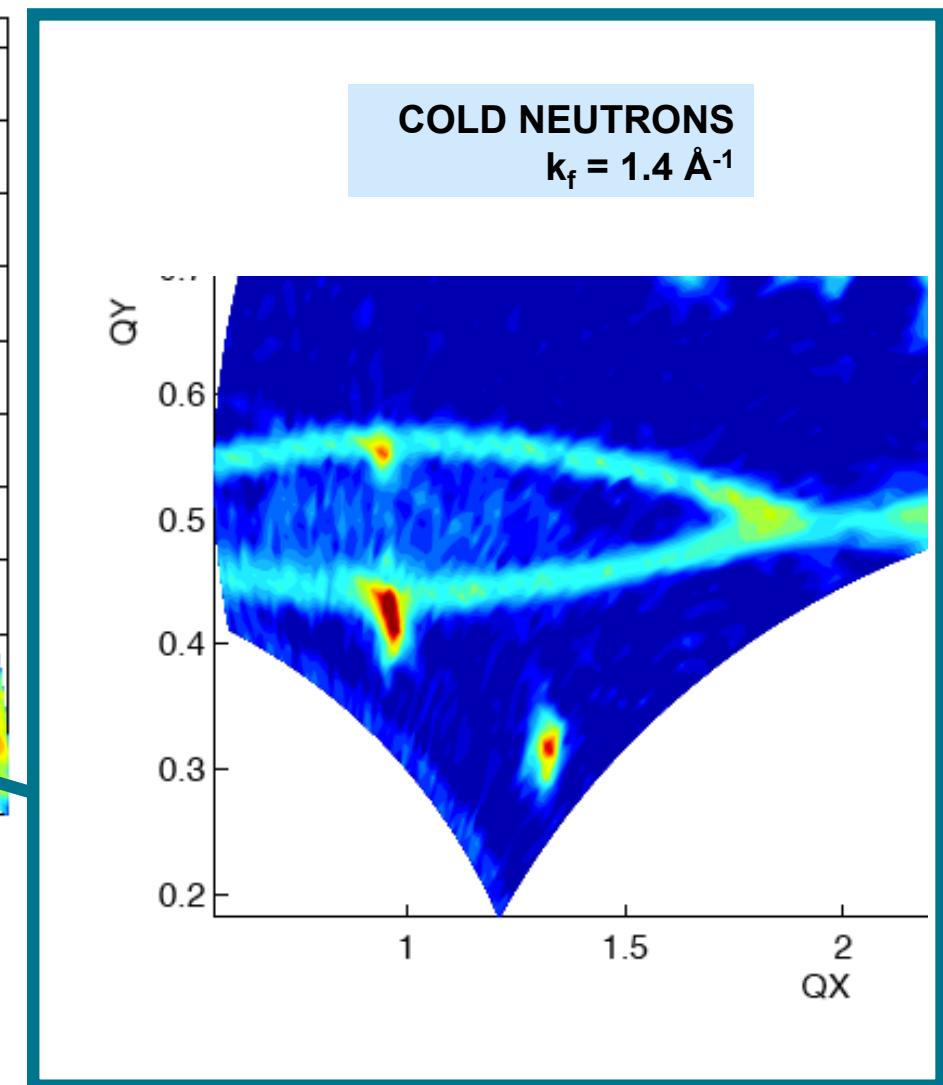
005

LANGEVIN

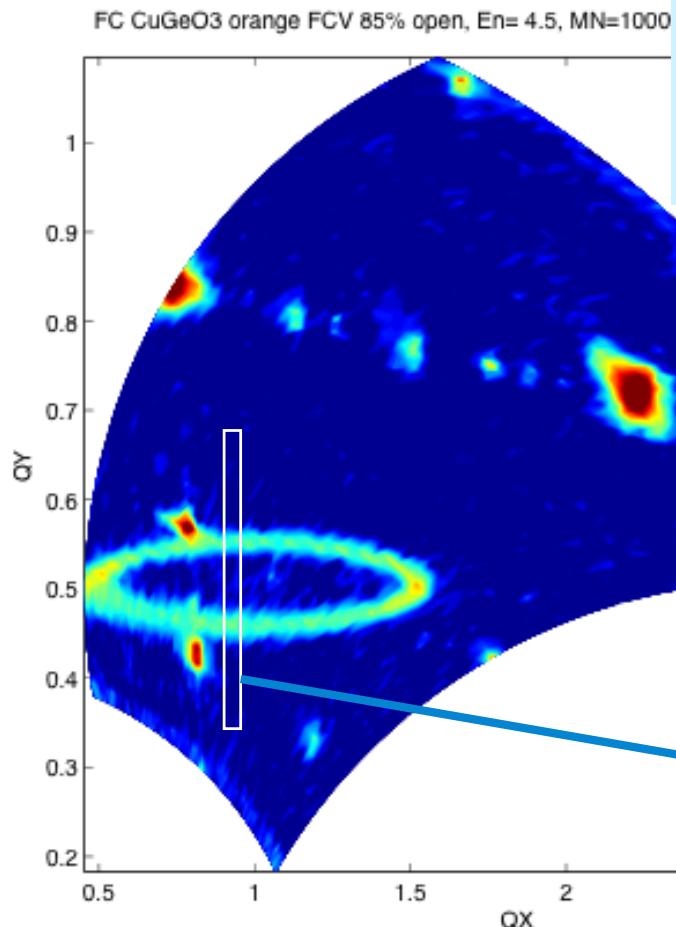
# CuGeO<sub>3</sub>



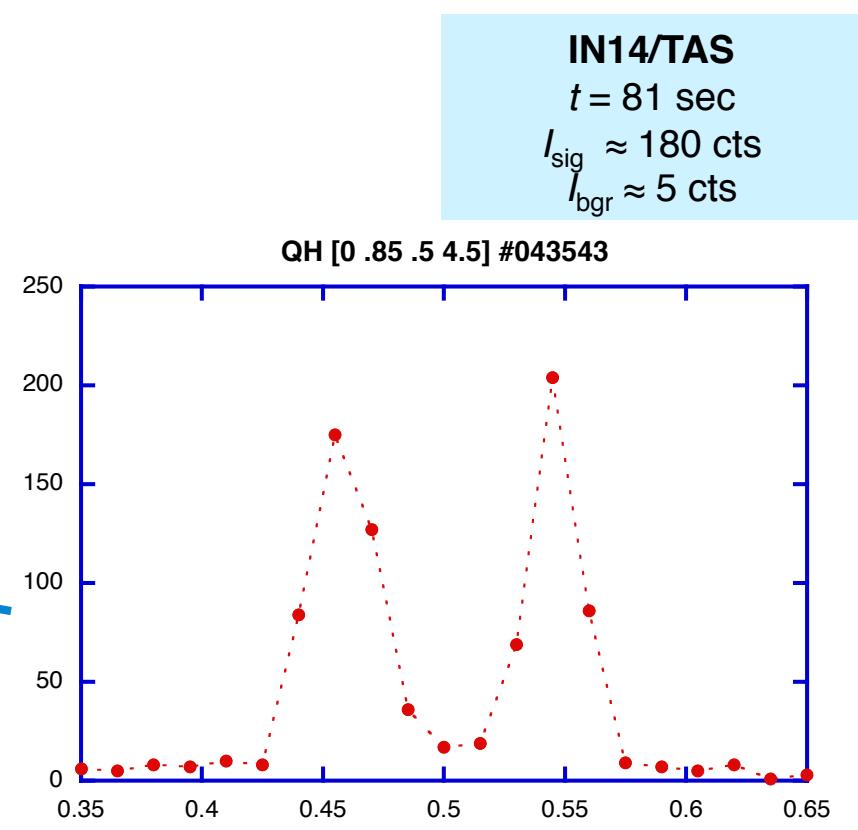
**THERMAL NEUTRONS**  
 $k_f = 1.4 \text{ \AA}^{-1}$



# CuGeO<sub>3</sub> with IN14/FC



**IN14/FC**  
 $t = 77 \text{ sec}$   
 $I_{\text{sig}} \approx 60 \text{ cts}$   
 $I_{\text{bgr}} \approx 5 \text{ cts}$



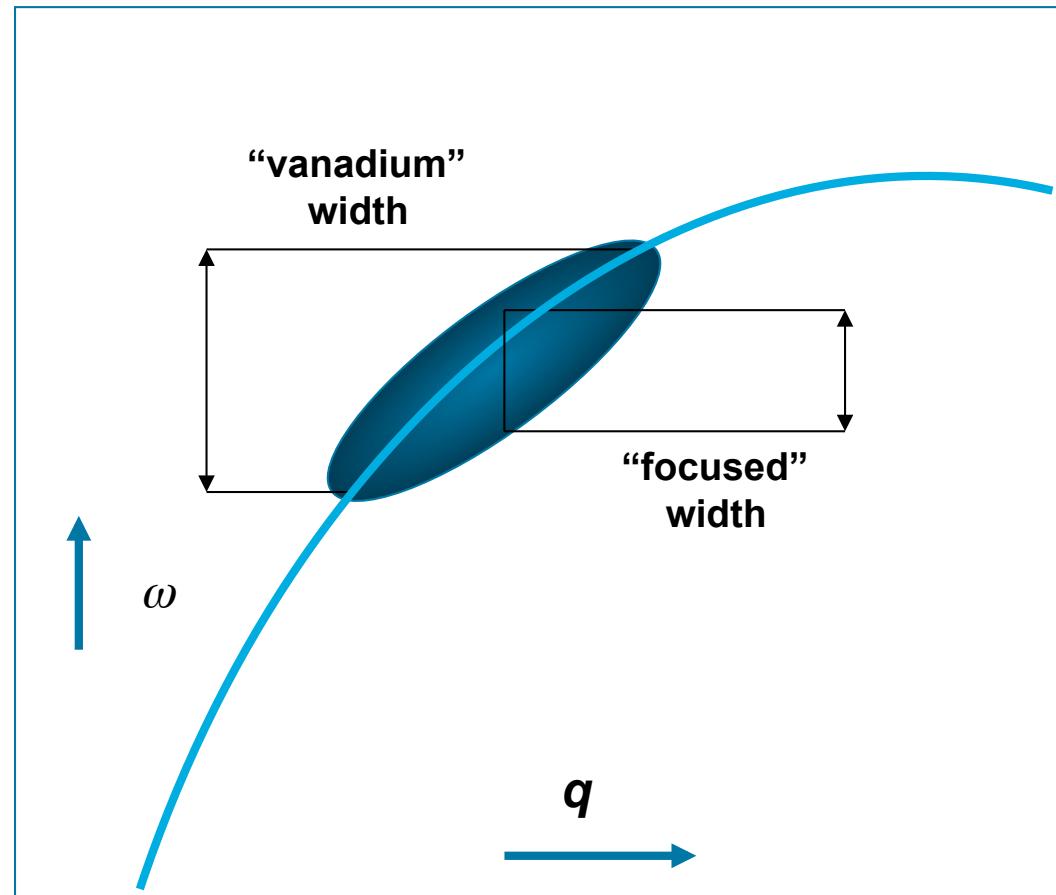
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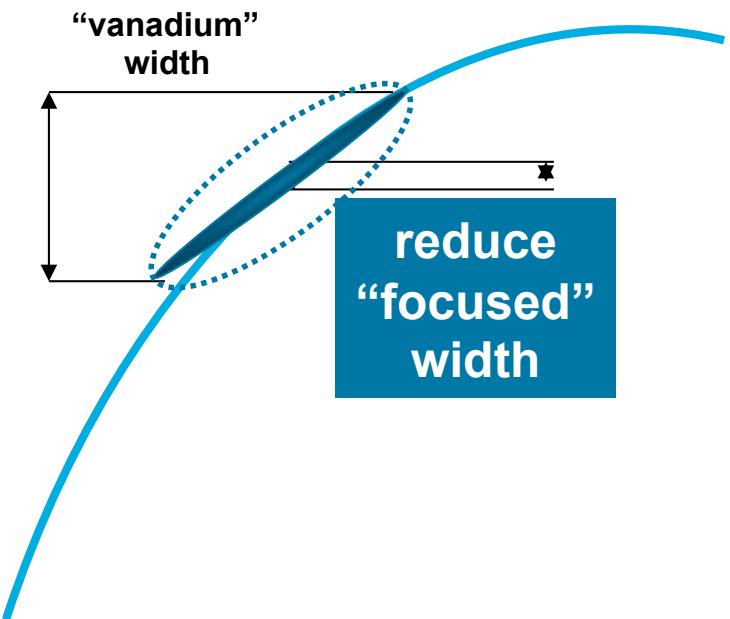
# Neutron TAS

## Neutron Three-Axis Spectrometers:

- access to large  $Q, \omega$  range
- energy resolution  $\Delta E/E \approx 5-10\%$
- efficient for  $\omega(q)$
- lacking resolution for  $I(q)$

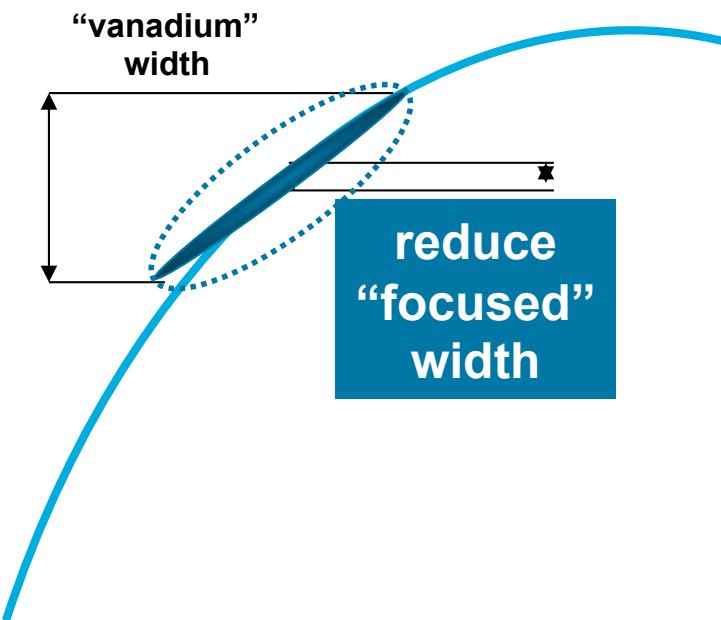


## TAS resolution

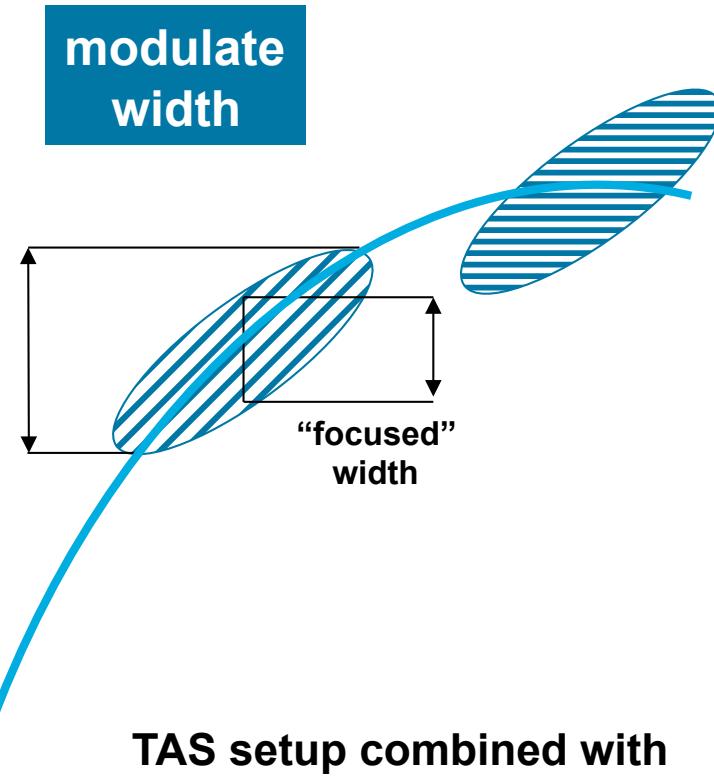


**normal TAS setup with  
perfect monochromator &  
analyzer crystals (Si, Ge)**

# TAS resolution

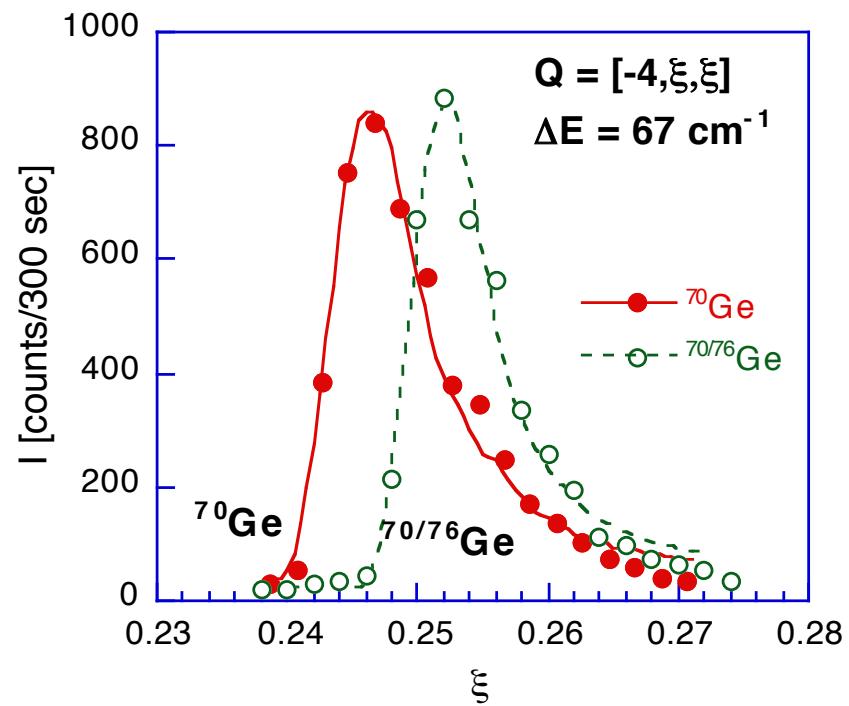


normal TAS setup with  
perfect monochromator &  
analyzer crystals (Si, Ge)



TAS setup combined with  
spin-echo  
(TOF Fourier technique)

# TASSE vers. high resolution TAS

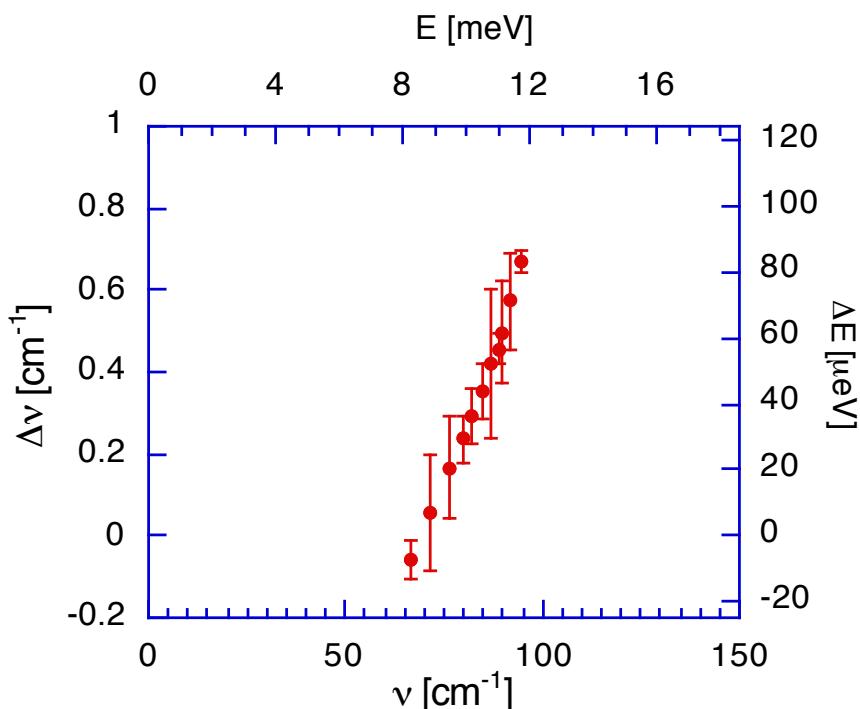


- same position accuracy
- HR TAS faster and easier
- TASSE better for FWHM

A. Goebel et al., Phys. Rev. B58 (1998) 10510

**IN20**

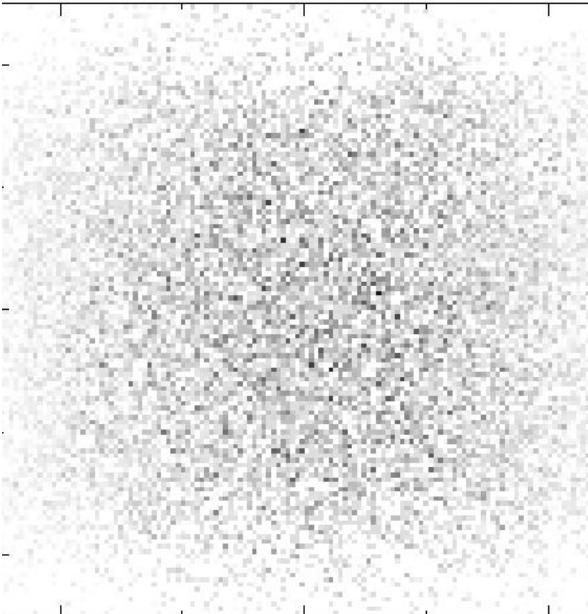
Si111/Si111  
 $R_M = R_A = 50 \text{ m}$   
sample volume  $< 0.5 \text{ cm}^3$



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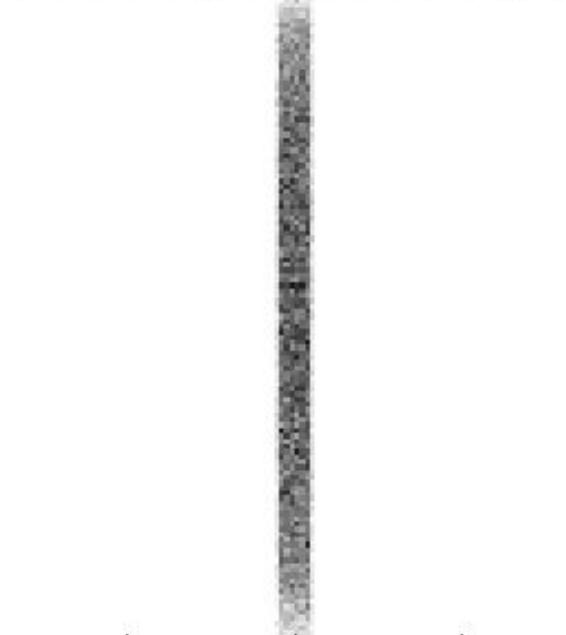
# Microfocusing crystal optics



**PG002 horizontal focus**  
**(RESTRAX ray-tracing)**



**Paris-Edinburgh**  
**High pressure cell**

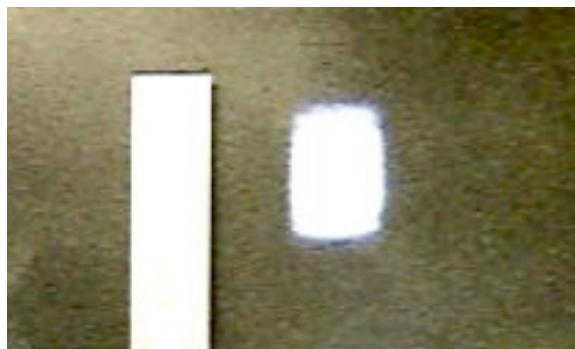


**Si111 horizontal focus**  
**(RESTRAX ray-tracing)**

# Si (111) microfocusing tests

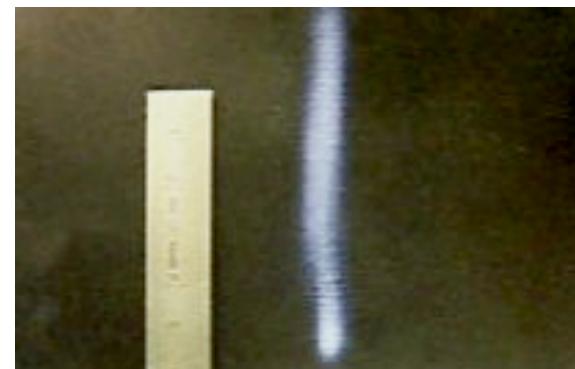
**IN20 (2009)**

*horizontal (focused)*



$D \approx 10 \text{ mm}$

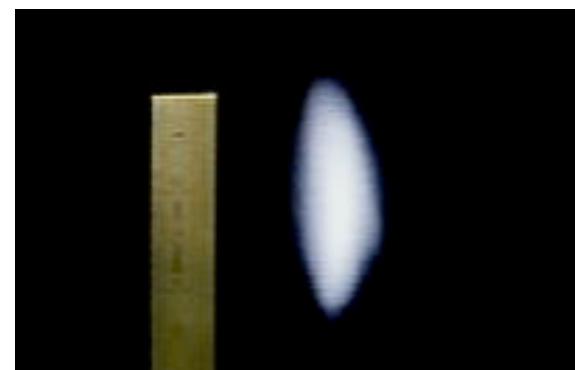
*vertical ( $D \approx 2 \text{ mm}$ )*



*defocused*

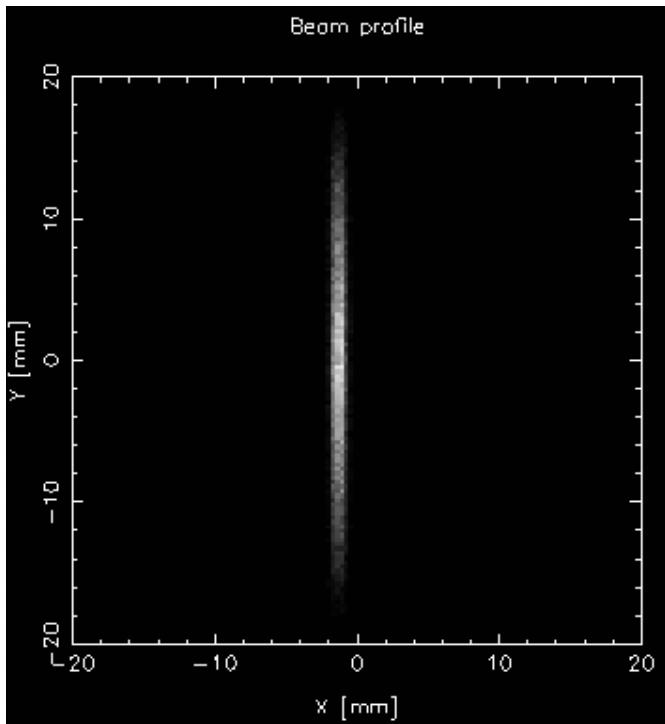


$D \approx 1 \text{ mm}$



*focused*

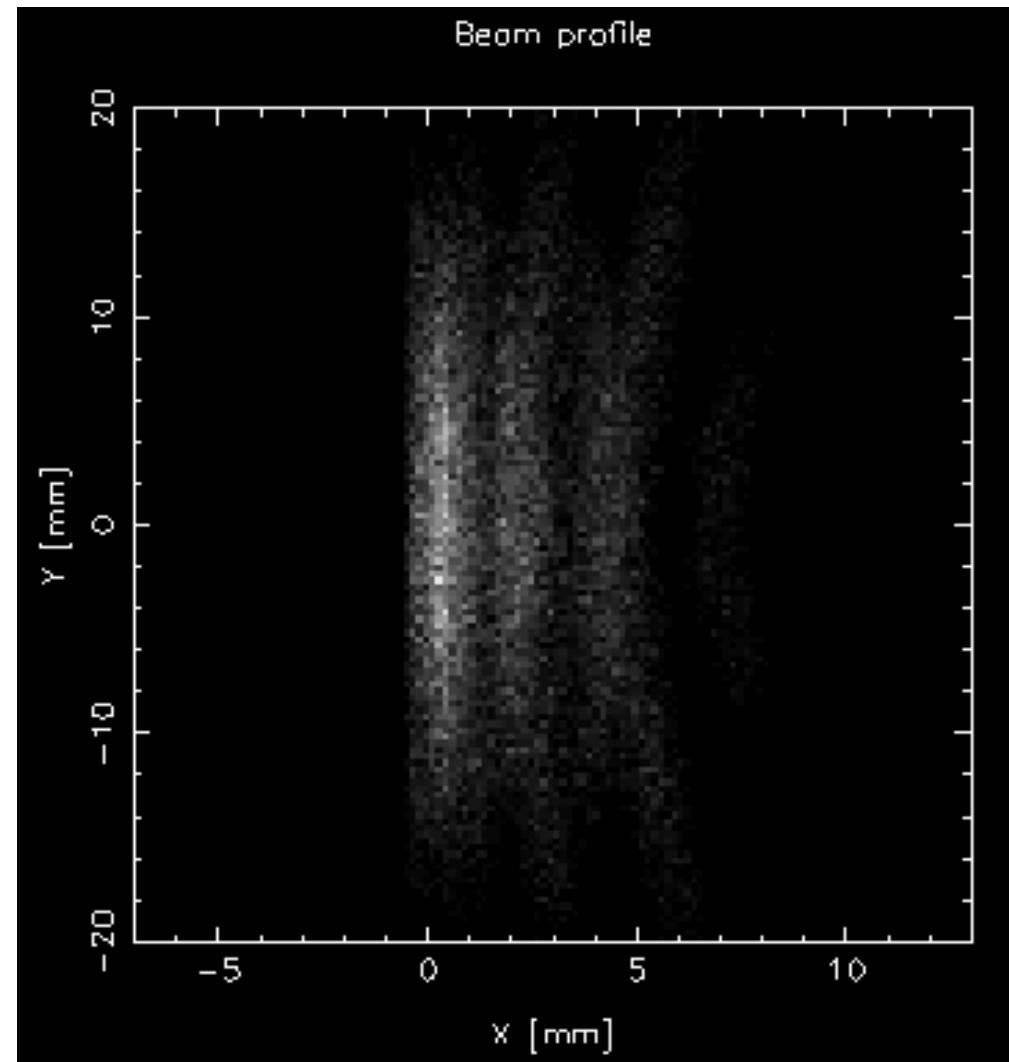
# Vertical focusing Si111



1 segment

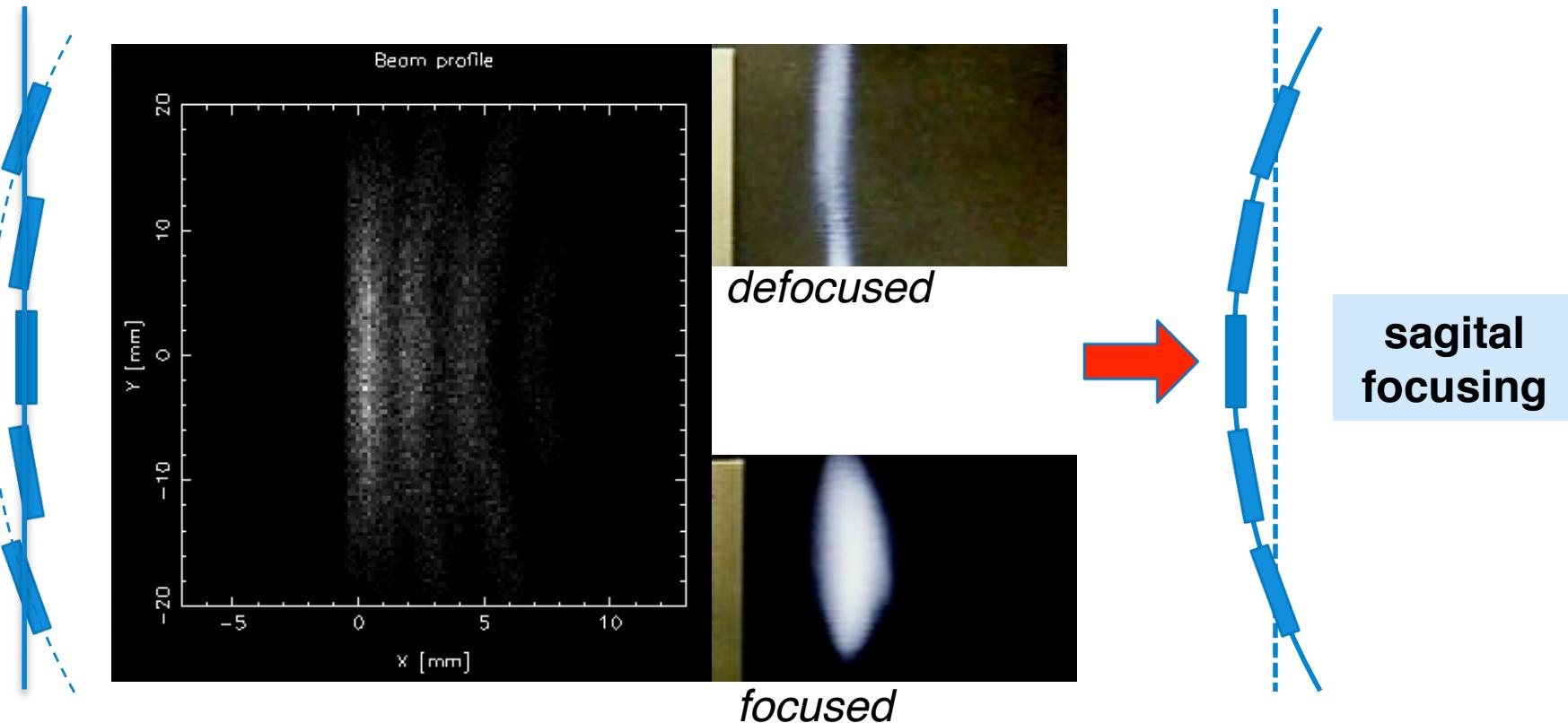
Restrax simulation  
**real space (X-Y)**

9 segments



# Vertical focusing aberrations

- standard TAS devices approximate cylindrical surface by a flat array of tilted slabs
- in high resolution mode aberrations due to variations in  $\theta_B$  become apparent
  - **develop a true sagitally focusing system**
  - **replace by a multichannel supermirror device**



# Concluding remarks

## Strong points

- deterministic behavior, sharp imaging
- almost rectangular reflection profiles
- absence of 2<sup>nd</sup> order contamination (Si, Ge, diamond)
- high transparency if Si (multicrystal alignments)

## Caveats

- needs precise manufacturing/alignment
- reveals irregularities of samples (sample assemblies)
- aberrations visible in high-resolution setups
- silicon cutting issues